



## 1. Introduction:

Quartz c-axis fabrics are widely used to infer flow kinematics (e.g. Lister and Hobbs, 1980; Price, 1985; Simpson and Schmid, 1983) as well as deformation temperature (e.g. Faleiros et al., 2016; Law, 2014) and mechanisms (e.g. Schmid and Casey, 1986; Stipp et al., 2002) of ductile shear zones in Earth's lithosphere. In terms of kinematics, the asymmetry of a c-axis fabric with respect to the shear zone coordinates is routinely used to infer sense of shear (e.g. Lister, 1977; Lister and Price, 1978; Lister and Williams, 1979; Menegon et al., 2008; Price, 1985). Fig.1 summarizes the standard models for this practice (e.g. Passchier and Trouw, 2005). Note we have adopted the convention of Lister (1977) to present the c-axis fabrics whereby the shear plane (C-foliation) is vertical east west and the shear direction (the Lc-lineation of Lin et al., 2007) is horizontal east. This convention differs from the one that orients the S-foliation east west and Ls-lineation horizontal east (e.g., Passchier and Trouw 2005). Natural quartz c-axis fabrics mainly comprise of single- and cross-girdles. In the coordinate system used here (Fig.1), the cross-girdle has its dominant girdle normal to the shear plane and the weaker girdle inclined antithetically to the shear sense (Fig.1a). Single c-axis girdles either incline antithetically to the shear sense or are nearly normal to the shear plane (Fig.1b, c). Point maxima within the girdles may lie at the periphery, the center, or intermediate positions between the periphery and the center. These are interpreted as reflecting the slip systems during deformation (e.g. Mainprice et al., 1986; Okudaira et al., 1995; Schmid and Casey, 1986; Simpson and Schmid, 1983). The models summarized in Fig.1 were based on numerical modeling of pure quartz aggregates using the Taylor-Bishop Hill model (e.g. Lister, 1977; Lister and Hobbs, 1980; Lister and Williams, 1979) and the visco-plastic self-consistent (VPSC) model (e.g. Morales et al., 2011; Nie and Shan, 2014; Wenk et al., 1989). Both model methods predicted that the c-axis girdle is antithetic or normal to the shear zone boundary in the coordinate system used here. Modeling has never generated girdles synthetically-inclined to the shear plane.

However, synthetically-inclined girdles have been observed in both natural shear zones (e.g. Keller and Stipp, 2011; Kilian et al 2011, Law et al., 2010; Little et al., 2016) and creep experiments of quartz aggregates (e.g. Heilbronner and Tullis, 2006; Kilian and Heilbronner, 2017). Heilbronner and Tullis (2006) suggested that the synthetical orientation is due to rotation with vorticity of earlier antithetic girdles as finite strain increases. But this is not supported by

any numerical modeling work with fairly large shear strains (up to 5, e.g. Jessell and Lister, 1990; Morales et al., 2011). Little et al (2016) suggested that general plane-strain flows may lead to synthetically-inclined c-axis girdles. The VPSC models of Takeshita et al. (1999) and Nie and Shan (2014) considered such flows but did not produce any synthetically-inclined c-axis girdles. However, their model did not consider cases where prism<a> and rhomb<a> slips are more significant than basal<a> slip. Through VPSC modeling, Keller and Stipp (2011) produced synthetically-inclined c-axis girdle with rhomb <a>, prism <a>, and prism <c> all active and all more significant than the basal <a> slip system. However, there are natural samples with synthetically-inclined c-axis girdles (e.g. Little et al 2016, Kilian et al 2011; Law et al 2010) that were clearly produced in the temperature range ~350-550°C much below that required for the activation of prism <c> slip (Toy et al., 2008). Furthermore, c-axis fabrics with both synthetically- and antithetically-inclined c-axis girdles have been observed in the same thin section (Kilian et al., 2011 their fig. 8, 9). Therefore, it is important to clarify if the combination of rhomb<a> and prism<a> slip, without prism <c>, can generate these synthetically-inclined c-axis girdles.

The standard models summarized in Fig.1 are based on the deformation of pure quartz aggregates under limited (mostly simple shearing) single-scale uniform flows (e.g. Lister et al., 1978; Lister and Hobbs, 1980; Lister and Williams, 1980; Keller and Stipp, 2011; Morales et al., 2014; Nie and Shan, 2014; Wenk et al., 1989). The spatial variation of quartz c-axis fabrics is a manifestation of heterogeneous deformation and, we suspect, is related to flow partitioning (e.g., Jiang, 1994a, 1994b; Lister and Williams, 1983) and strain buildup in rheologically distinct domains. Most natural mylonites are made of polyphase minerals in which the microscale rheology varies from one domain to another which facilitates flow partitioning. In fact, many authors have referred to flow partitioning qualitatively (e.g., Killian et al. 2011; Garcia Celma, 1982; Jerabek et al., 2007; Larson et al., 2014; Law, 1987; Lister and Price, 1978; Passchier, 1983; Pauli et al., 1996; Peternell et al., 2010) to explain observed c-axis fabric variations.

In this contribution, we apply a multiscale numerical modeling approach to quantitatively investigate the consequence of flow partitioning on the development of quartz c-axis fabrics and compare our modeling results with natural and experimental observations. Specifically, we use the mylonite thin-section photomicrograph of Killian et al (2011, Fig.8 there) as a model and

seek to understand if flow partitioning can produce the observed variation in quartz c-axis fabrics.

## 2. Approach

Fig.2a shows the thin-section photomicrograph from Kilian et al (2011). The rock is comprised of quartz domains (from which c-axis fabrics were presented), feldspar porphyroclasts, and a matrix of fine-grained quartz, feldspar, and mica. According to Kilian et al. (2011), the microstructure was produced in a simple shearing flow which is consistent with the geometric pattern of the foliation (Ramsay, 1980; Ramsay and Graham, 1970). In this investigation, we consider plane-strain general shearing flows as the bulk flow field to understand how the partitioning of the bulk flow into rheologically distinct quartz domains may affect the c-axis fabrics in those domains. Our method here also applies to any 3D general shearing flows (e.g. Jiang and Williams, 1998).

In the coordinate system used in this investigation (Fig.2c), a general plane-strain general shearing flow is defined by the following Eulerian velocity gradient tensor:

$$\mathbf{L} = \begin{pmatrix} \dot{\epsilon} & \dot{\gamma} & 0 \\ 0 & -\dot{\epsilon} & 0 \\ 0 & 0 & 0 \end{pmatrix} \quad (1)$$

where  $\dot{\gamma}$  is the shear strain rate for the simple shearing component and  $\dot{\epsilon}$  is the strain rate parallel to the X-axis. The flow in Eq.1 corresponds to a kinematic vorticity number

$$W_k = \frac{\dot{\gamma}}{\sqrt{4\dot{\epsilon}^2 + \dot{\gamma}^2}} \quad (\text{Jiang and White, 1995; Li and Jiang, 2011; Truesdell, 1953}).$$

We consider the variation of the flow by varying  $W_k$  from 0 to 1.

The progression of the finite strain is measured by a strain intensity defined as  $\rho =$

$$\sqrt{\left(\ln \frac{s_1}{s_2}\right)^2 + \left(\ln \frac{s_2}{s_3}\right)^2} \quad \text{where } s_1, s_2, s_3 \text{ are the three principal stretches } (s_1 > s_2 > s_3) \text{ of the}$$



finite strain ellipsoid (e.g., Yang et al., 2019). In the event of simple shearing ( $W_k = 1$ ), the strain can also be measured by the shear strain  $\gamma$ . The relation between  $\gamma$  and  $\rho$  in simple shearing situation is shown in Fig.3.

We use the Visco-Plastic-Self-Consistent (VPSC) model, originally due to Molinari et al (1987) and further developed by Lebensohn and Tome (1993), to simulate the quartz c-axis fabric development in a flow field. Specifically, VPSC7 (Lebensohn and Tomé, 2009) for Windows is used in this investigation. We track the c-axis evolution of 500 quartz crystals whose initial orientations are randomly distributed in 3D space (Jiang 2007b). The initial shapes of grains are equant. The crystal shapes evolve with strain and the grain fragmentation scheme of Beyerlein et al. (2003) is used to limit the aspect ratio of quartz grains to mimic some effect of dynamic recrystallization. The VPSC output of c-axis data are plotted using the MTEX toolbox (Bachmann et al., 2010). Since we are concerned with variation in quartz c-axis fabrics, we present pole figures of the (0001) directions here. Pole figures of other crystal directions are provided in the supplementary material.

The relative activity of slip systems is modulated in VPSC by their relative critical resolved shear stress (CRSS). A lower CRSS corresponds to higher activity. For quartz, the CRSS for a slip system is largely temperature dependent with basal  $\langle a \rangle$  slip occurring at about 350-500°C, rhomb  $\langle a \rangle$  and prism  $\langle a \rangle$  slip at  $\sim 500$ -600°C, and prism  $\langle c \rangle$  slip at  $> 650^\circ\text{C}$  (e.g. Toy et al., 2008). Table 1 summarizes the slip system combinations used in this study based on previous work (e.g. Lister and Paterson, 1979; Morales et al., 2014; Wenk et al., 1989). They are labelled as Model-A to F. Morales et al (2014) and Keller and Stipp 2011) made a distinction between rhomb (+) slip  $\{r\}$  in  $\langle a \rangle$  direction and rhomb (-) slip  $\{z\}$  in  $\langle a \rangle$  direction in their modelling works. This distinction may be important in regime 1 dislocation creep (Hirth and Tullis, 1992) but it is not necessary for our work here on the effect of flow partitioning.

In the single scale case, the flow field defined in Eq.1 is used in the VPSC directly for the simulation of the resulting quartz c-axis fabric development. In the multiscale case, Eq.1 defines the bulk macroscale flow which must be partitioned into different rheologically distinct quartz domains (Figs. 2b and d) before the quartz c-axis fabric development in those domains can be simulated with VPSC. We use the self-consistent Multi Order Power Law Approach (MOPLA) (Jiang and Bentley 2012; Jiang 2016, 2014; Qu et al., 2016; and Lu 2020) to obtain the

partitioned flow fields in quartz domains. The multiscale approach can be illustrated using the thin section sample in Fig.2a as follows: Quartz domains, feldspar porphyroclasts, and mica seams are referred to as Rheologically Distinct Elements (RDEs). The sample as a whole was subjected to a *macroscale* flow field like Eq.1. In micromechanics terms, the macroscale flow is the bulk flow *averaged* on a Representative Volume Element (RVE) which represents the mineral assemblage for the sample. It is reasonable to regard the thin section as a section of the RVE for the macroscale flow. The *microscale* flow field in each constituent RDE such as a quartz domain is distinct and differs from the macroscale flow, because each RDE is unique in its rheology, shape, and orientation (Eshelby, 1957; Mura, 1987; Jiang 2014, 2016). Clearly, it is the microscale (or partitioned) flows in quartz domains that are responsible for the quartz c-axis fabric development.

In MOPLA, an RDE is regarded as an ellipsoidal Eshelby inhomogeneity embedded in and interacting with the macroscale material (Fig.2d). The rheology of the latter is approximated by a homogeneous effective medium (HEM) and obtained from the rheologies of the constituent RDEs from a set of homogenization equations. The microscale or partitioned flows in an RDE is related to the macroscale flow, which is assigned by Eq.1, by a set of partitioning equations (for details, see Jiang 2014, 2016). The partitioning and homogenization equations are solved simultaneously to obtain the partitioned flow fields and the macroscale rheology. As the relevant rheology for mylonites are power-law viscous (Kohlstedt et al., 1995), the MOPLA formulation adopts a linearization approach (Lebensohn and Tomé, 1993; Molinari et al., 1987) where linearized viscosities such as tangent viscosities are used in the formulation. As we are concerned with quartz c-axis fabric development, we specifically use MOPLA to calculate the partitioned flow fields within quartz domains.

We consider two situations for the rheology of the quartz RDEs and HEM. In the first, both the RDEs and the HEM are isotropic. The rheological contrast between an RDE and the HEM is reduced to an effective viscosity ratio  $r$  between the RDE to the HEM. We do not consider the rheological anisotropy development in HEM as a result of fabric buildup with strain because anisotropic rheological response of the constituent RDEs are not available. Because of power-law rheology,  $r$  varies with time. We thus consider a range of constant  $r$  values. The quartz aggregates in Fig.2a all have convex shapes with surrounding matrix material wrapping

around them, suggesting that quartz RDEs were rheologically stronger ( $r > 1$ ) than the ambient HEM. But  $r$  cannot be too high (e.g.,  $r > 10$ ) because of power-law rheology or the quartz RDEs would behave like rigid clasts (Jiang, 2007a; Xiang and Jiang, 2013) with no c-axis fabric formation. We consider the situations of  $r$  being 2, 5, and 10. The situation of  $r = 0.5$  is also considered here for comparison to show what the c-axis fabrics might be like if quartz RDEs were mechanically weaker than the ambient medium.

In the second situation, we consider isotropic RDEs in a HEM of simple planar anisotropy which approximates a foliated and/or layered material like natural mylonites. The rheology of such a HEM can be characterized by two distinct viscosities:  $\eta_n$ , the normal viscosity for the resistance to pure shearing along and perpendicular to the layering, and  $\eta_s$  the shear viscosity measuring the resistance to shearing parallel to the layering (e.g. Jiang, 2016; Fletcher, 2009; Johnson and Fletcher, 1994). The strength of anisotropy is measured by the ratio  $m$  of  $\eta_n$  to  $\eta_s$ . For foliated and layered rocks,  $m > 1$  (Treagus, 2003). The rheological contrasts between the isotropic quartz RDE and HEM can be defined by the following two parameters: the ratio,  $r_{eff}$ , between the viscosity of the RDE to  $\eta_n$  and  $m$ . In such case, the effective viscosity of the RDE is simply given by  $r_{eff}\eta_n$ . Similar to the isotropic cases, we consider  $r_{eff}$  being 0.5, 2, 5, and 10. Fig. 4 shows the geometric relation between the flow field and the plane of anisotropy. Macroscale flow is simple shear with shear plane parallel to X-Z plane.

To cover the shape variation of quartz RDEs, we considered three reference initial shapes: prolate (5:1:1), oblate (5:5:1), sphere (1:1:1) and initial triaxial RDEs with long and short semi-axial length fixed to 5 and 1 respectively and intermediate semi-axial length ranging from 2-4. These initial RDEs will deform into various possible triaxial shapes in nature. The initial orientations of the RDEs are defined by spherical angles (Jiang, 2007b, 2007a) which are randomly assigned.

The partitioned flow for a quartz RDE computed from MOPLA is used as the input flow field to simulate the quartz c-axis fabrics in that RDE through the VPSC model. Because the partitioned flow field in any given RDE is non-steady as the RDE continuously changes shape and orientation during deformation, the coupled computation between MOPLA and VPSC is carried out as follows: With a given macroscale flow field defined in Eq.1, we use MOPLA

algorithm implemented in MATLAB (Lu, 2020; Jiang, 2016, 2014, 2007a; Jiang and Bentley, 2012; Qu et al., 2016) to calculate the partitioned flow in every quartz RDE, which is expressed as a velocity gradient tensor. We export to a data file the RDE velocity gradient tensor for every prescribed macroscale strain increment until a pre-set macroscale finite strain is reached. This data file is then used as input flows into the VPSC code to calculate the c-axis fabric evolution within the RDEs as macroscale strain increases until the set magnitude. The velocity gradient tensor files for all the RDEs are available in the supplementary material.

### 3. Results:

As mentioned above, we have simulated quartz c-axis fabric development in three situations. The first is in homogeneous macroscale plane-strain general shearing flows, without flow partitioning, with quartz slip system combinations that have not been covered by previous studies. The second situation is when rheologically distinct quartz domains are within an isotropic HEM, and the third is when the quartz domains are within a HEM of planar anisotropy.

#### 3.1 Quartz c-axis fabric development in homogeneous plane-strain general shearing flows

Figs.5 and 6 present quartz c-axis fabrics produced for models-A-F (rows) under uniform macroscale flows of plane-strain general shearing from  $W_k = 0$  to  $W_k = 1$  (columns), at macroscale strain states  $\rho = 2$  and 6.

In pure shearing ( $W_k = 0$ ), for models-A, C, and D, peripheral c-axis maxima form and remain at the maximum shortening direction regardless of strains (Figs. 5a,d, m,p,6a,d). For model-B, a cross-girdle (Figs 5g) form at  $\rho = 2$ , with its central segment lying along the maximum shortening direction. A single girdle is produced lying along the maximum shortening direction (Figs. 5j) at  $\rho = 6$ . For models-E and F, peripheral maxima develop and remains at the maximum stretching direction at both strain states (Figs. 6g, j, m, p).

When  $0 < W_k \leq 1$ , for models-A, C, and D and at  $\rho = 2$ , peripheral c-axis maxima are developed inclining antithetically to the shear sense (Figs. 5b, c, n, o, 6b, c). As  $W_k$  increases, the angle between the peripheral c-axis maxima and the shear plane normal increases. The peripheral

maxima rotate with macroscale vorticity as finite strain increases but do not pass the shear plane normal (Figs. 5f, l, q, r, 6f). For model-B, a cross-girdle, with its central segment lying normal to the shear plane, is formed at  $\rho = 2$  (Figs 5 h, i). At  $\rho = 6$ , this cross-girdle becomes a single girdle lying normal to the shear plane (Figs. 5 k, l). In some general shear ( $0 < W_k < 1$ ) cases, the c-axis girdles can be slightly synthetically-inclined at  $\rho = 6$  (Figs. 5e,k,6e), but the angle between the peripheral c-axis maxima and the shear plane normal is small ( $< 10^\circ$ ). For models-E and F, the c-axis peripheral maxima are synthetically-inclined near the shear direction at  $\rho = 2$ . The angle between the peripheral c-axis maxima and the shear direction increases with  $W_k$  (Figs.6h, i, n, o). The c-axis peripheral maxima rotate with vorticity toward the shear direction as the finite strain increases (Figs.6 k, l, q, r).

### 3.2 *Quartz c-axis fabric development in quartz domains embedded in an isotropic HEM*

Since models-A, B, C and D all produce similar c-axis fabrics with c-axis girdles antithetically-inclined or nearly normal to the shear plane, we only present results for model-A for multiscale deformation. Figs. 7 and 8 report ten results of c-axis fabrics produced in quartz RDEs of varying initial shapes, orientations, and viscosity ratio  $r$  under a range of macroscale flow fields. These results can be summarized as follows: When  $0 < W_k \leq 1$ , c-axis girdles are always antithetically-inclined (Figs.7a,b,d,e,g,h, j, k, m, n, 8a,g,j,m) regardless of initial conditions of RDEs up to the finite strain  $\rho \sim 4$ . With increase in finite strain, the girdles rotate with bulk vorticity (rows of Figs.7 and 8) but do not pass the shear plane normal unless  $r \geq 5$  (Figs. 7-i, l, o, 8-b, c, i, l, o). If the RDEs were weaker than the HEM ( $r = 0.5$ ), c-axis girdles remain close to normal (Figs. 7-c) to the shear plane even at very high finite strains ( $\rho \sim 6-7$ ). In pure shearing, a cross girdle is produced at  $\rho \sim 2$  which becomes a single girdle at  $\rho \sim 6$  with peripheral c-axis maxima always parallel to the maximum shortening direction (Figs.8-d, e, f).

### 3.3 *Quartz c-axis fabric development in quartz domains embedded in a planar anisotropic HEM*

Figs. 9a-o report c-axis fabrics developed in quartz RDEs of varying initial shapes, orientations, and viscosity ratio  $r_{eff}$ , embedded in a planar anisotropic HEM of anisotropic strength  $m$  as described in Section-2. These results can be summarized as follows: The c-axis girdles are always antithetically-inclined (Figs.9a,d,g,j,m) at  $\rho \sim 2$  regardless of  $W_k$ ,  $r_{eff}$ ,  $m$ , initial orientations and shapes of RDEs. The girdles rotate with bulk vorticity as  $\rho$  increases (rows of

Figs.9) but do not pass the shear plane normal unless  $r_{eff} \geq 2$  (Figs. 9i, l, o). If RDEs were weaker ( $r_{eff} = .5$ ), c-axis girdles are close to normal (Figs. 9c, f) to the shear plane even at high finite strains ( $\rho \sim 6-7$ ).

#### 4. Discussion:

In Section 2, we argued that  $r$  must be greater than 1 from microstructures (Fig.2a) but not so high that the quartz RDEs do not develop enough internal strain for c-axis fabric formation. In our modeling, we considered the range  $r$  between 2 and 10. Our modeling results show that, with basal<a>, rhomb<a>, and prism<a> slips, c-axis girdles in a quartz RDE always develop at an antithetical orientation initially. But the girdles rotate with vorticity as the macroscale finite strain increases. If the RDE is sufficiently strong ( $r \geq 5$  in isotropic HEM case and  $r_{eff} \geq 2$  in planar anisotropic HEM case), the girdles will rotate pass the shear zone normal and lie in the synthetical sector at high strains ( $\rho \sim 6$ , Figs. 7-i, l, o, 8-b, c, i, l, o, 9-i, l, o). Our modeling results show that for synthetically-inclined girdles,  $r$  should be between 5 and 10. These results are consistent with the strain gradient of the thin section (Fig.10, based on Killian et al., 2011, Fig.9 there). The antithetically-inclined c-axis peripheral maxima (yellow in Fig.10) correspond to a lower strain and synthetically-inclined c-axis peripheral maxima (red in Fig. 10) to a higher strain.

Despite the variability of the partitioned flow field in quartz RDEs, we found out that the microscale vorticity in every quartz RDE still has the same sense as the macroscale vorticity. Fig.11 shows the dot product of the unit vector  $\hat{\omega}$  parallel to the vorticity in a RDE and the unit vector  $\hat{\Omega}$  parallel to the macroscale vorticity.  $\hat{\omega} \cdot \hat{\Omega}$  is positive for all quartz RDEs (a few selected ones are shown in Fig.11). In other words, there is no vorticity sense reversal in any quartz RDEs. This implies that the volume-weighted average of microscale vorticity vectors is parallel to the macroscale vorticity vector. Therefore, the crystallographic vorticity axis (CVA) analysis (Giorgis et al., 2017; Michels et al., 2015) can still be used in every quartz RDE and then the averaged microscale vorticity axes represent the macroscale vorticity axis.

Synthetically-inclined c-axis girdles have also been reported in some creep experiments on pure quartz aggregates with similar slip systems (basal<a>, rhomb<a>, and prism<a>) at high finite strains (Heilbronner and Tullis 2002, 2006). Although Keller and Stipp (2011) obtained

synthetically-inclined c-axis girdles by including prism  $\langle c \rangle$  slip in their VPSC models, we have further confirmed that without prism  $\langle c \rangle$  slip (our model-D), synthetically-inclined c-axis girdles cannot be produced (Figs. 5a-f), unless flow partitioning is considered. We suspect that some degree of heterogeneous strain and therefore partitioned flow was responsible for such c-axis girdle orientations. Heilbronner and Tullis (2006) themselves suggested that different domains of polycrystal aggregates might exhibit different viscosities, which could have facilitated partitioning of the flow among different domains.

Li and Jiang (2011) raised issues with the practice of vorticity estimation using rigid porphyroclasts under the assumption of steady-state homogeneous flow histories. The significance of flow partitioning as demonstrated by our modeling based on microstructures of natural mylonites raises further issues with using quartz c-axis fabrics to estimate the (macroscale) vorticity (e.g. Vissers, 1989; Wallis, 1995, 1992; Xypolias, 2009; Law, 2010). First, where quartz c-axis fabrics have resulted from partitioned flow, the steady-state flow assumption is invalid. As we have shown, there is a distinct microscale vorticity history, not a constant vorticity number, in every quartz RDE, which cannot be determined from the final c-axis fabric. In principle, the microscale vorticity in a quartz RDE does not have a simple relation to the macroscale vorticity. Second, even for the single-scale case where no flow partitioning is considered, our modeling demonstrates that the assumption commonly used in vorticity determination that the dominant c-axis girdle is perpendicular to the shear plane is not always valid.

Peripheral c-axis maxima close to the shear direction have commonly been taken to reflect prism  $\langle c \rangle$  slip (Passchier and Trouw, 2005). Our modeling suggests that they can also be significantly rotated peripheral basal  $\langle a \rangle$  maxima from certain quartz RDEs (Fig.8c). Larson et al (2014) have reported a possible example of this. They presented peripheral c-axis maxima close to the shear direction in a temperature condition much below that required for the activation of prism  $\langle c \rangle$  and used the concept of flow partitioning to explain their observation. Our modeling lends support to this explanation.

## 5. Conclusions:

The co-existence of both synthetically-inclined and antithetically-inclined quartz c-axis girdles in a single thin section can be explained by flow partitioning at the thin section scale. The antithetically-inclined girdles correspond to relatively low finite strains and the synthetically-inclined girdles to high finite strains.

Although the microscale flow fields vary from one quartz RDE to another and are distinct from the macroscale flow, the sense of vorticity in every quartz RDEs remains the same as the macroscale vorticity.

Because of flow partitioning, it is not possible to estimate the vorticity number of the macroscale flow from quartz c-axis fabrics. But, it is still possible to obtain the macroscale vorticity axis by averaging the microscale vorticity axes from quartz RDEs. The latter can be obtained through the crystallographic vorticity axis analysis .

As a result of partitioned flows, the dominant quartz c-axis girdle can lie antithetical, normal, synthetical to the shear plane. The basal  $\langle a \rangle$  peripheral maxima may end up lying close to the shear direction at high macroscale strains. Caution should be taken not to misinterpret these peripheral maxima as reflecting prism  $\langle c \rangle$  slip.

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**Figure Captions:**

**Figure 1:** Current models for interpreting quartz c-axis fabrics at low to medium temperatures ~350-550°C presented in the shear zone coordinate system. The sense of shear is dextral. Cross-girdle and single girdle c-axis fabrics are commonly observed. (a) c-axis cross girdle pattern with one girdle normal to the shear plane while the other antithetic to the shear sense. (b) a single c-axis girdle with Y-maxima, inclined either antithetically to the shear sense or normal to the shear plane. (c) a single c-axis girdle inclined either antithetically to the shear sense or normal to the shear plane. The c-axes near the periphery, the center, and in between are interpreted to reflect basal<a>, prism<a>, and rhomb<a> slips respectively.

**Figure 2:** Illustration of the multiscale approach used in this paper. (a) The thin section photomicrograph of a natural mylonite from Kilian et al. (2011) used as a model. (b) Sketch of (a). The thin section can be viewed as a 2D section of the representative volume element (RVE) for the shear zone material, which is composed of quartz domains, feldspar porphyroclasts, and mica seams in a fine-grained matrix. The quartz domains and feldspar clasts are referred to as Rheologically Distinct Elements (RDEs). We are concerned with partitioned flows in quartz RDEs in this paper. (c) Coordinate system to define the macroscale flow field used in modeling investigation (d) Each quartz RDE is regarded as a heterogeneous Eshelby inclusion embedded in the composite shear zone material that is idealized as the Homogeneous Equivalent Matrix (HEM). Microscale fields ( strain rate  $\mathbf{\epsilon}$  , and vorticity  $\mathbf{w}$  ) are related to respective macroscale fields (  $\mathbf{E}$  and  $\mathbf{W}$  ) by partitioning equations, where  $\mathbf{A}$  is the strain partitioning tensor,  $\mathbf{S}$  and  $\mathbf{\Pi}$  are respectively the 4<sup>th</sup>-order symmetric and anti-symmetric Eshelby tensors (Jiang 2014).



**Figure 3:** The macroscale finite strain of the shear zone, measured by the strain intensity  $\rho$ , as a function of computation steps for varying  $W_k$ . The increase of the shear strain  $\gamma$  in the simple shear case ( $W_k = 1$ ) is also plotted for comparison.

**Figure 4:** Geometric relation between the macroscale flow field and anisotropy plane in the planar anisotropic HEM.

**Figure 5:** C-axis fabrics in single-scale deformation with varying  $W_k$  from pure shearing to simple shearing and for models-A-C. The final strain intensity is between  $\rho = 2$  and 6. The pole densities are contoured in multiples of uniform distribution.

**Figure 6:** Same as Figure 5 but for models-D-F.

**Figure 7:** C-axis fabrics in selected quartz RDEs in an isotropic HEM under simple shearing ( $W_k = 1$ ). (a)-(o) are the resultant c-axis fabrics developed. The first column presents the initial conditions for the RDEs [  $r$  : viscosity ratio of the RDE to HEM, initial shape defined by semi-axes of the RDE ( $a_1 : a_2 : a_3$ , where  $a_1 \geq a_2 \geq a_3$ ), and initial orientation given by spherical angles ( $\theta_1, \Phi_1, \theta_2$ ) for general RDEs or ( $\theta, \Phi$ ) for spheroidal RDEs] . Each row presents the results for the RDE as the macroscale strain increases.

**Figure 8:** Same as Figure 7 except that the macroscale flow is plane strain general shearing ( $0 < W_k \leq 1$ ).

**Figure 9:** C-axis fabrics in selected quartz RDEs in a HEM with planar anisotropy under simple shearing flow ( $W_k = 1$ ) (a)-(o) are the resultant c-axis fabrics developed. The first column presents

the initial conditions for the RDEs [ $r_{eff}$  - viscosity ratio of the RDE to HEM's  $\eta_m$ , initial shape defined by semi-axes of the RDE ( $a_1 : a_2 : a_3$ , where  $a_1 \geq a_2 \geq a_3$ ), initial orientation given by spherical angles ( $\theta_1, \Phi_1, \theta_2$ ) for general RDEs or ( $\theta, \Phi$ ) for spheroidal RDEs, and anisotropic strength  $m$ ]. Each row presents the results for the RDE as the macroscale strain increases.

**Figure 10:** Sketch of the thin-section sample showing c-axis fabric variation in quartz domains (based on Kilian et al., 2011, Fig. 9 there). C-axis fabrics comprise of peripheral c-axis maxima that are antithetically-inclined (yellow), nearly normal (blue), and synthetically-inclined (red) to the shear plane. The sense of shear is dextral. The arrow at the top-left shows the direction of increase in strain gradient.

**Figure 11:** Plot of dot product  $\hat{\omega} \cdot \hat{\Omega}$  with increasing macroscale strain  $\rho$ , for 50 RDEs with random initial shapes and orientations,  $r = 0.5, 2$  and  $5$  (rows) and  $W_k = 1$  and  $.75$  (columns).  $\hat{\omega}$  and  $\hat{\Omega}$  are, respectively, the unit vectors parallel to the microscale vorticity vector in an RDE and the macroscale vorticity vector.  $\hat{\omega} \cdot \hat{\Omega}$  is close to unity for the RDEs. Therefore, the microscale vorticity vectors are always of the same sense as and nearly parallel to the macroscale vorticity vector.

**Table 1:** Quartz slip systems and the relative CRSS values for different models used in VPSC simulation.

Figure 1.

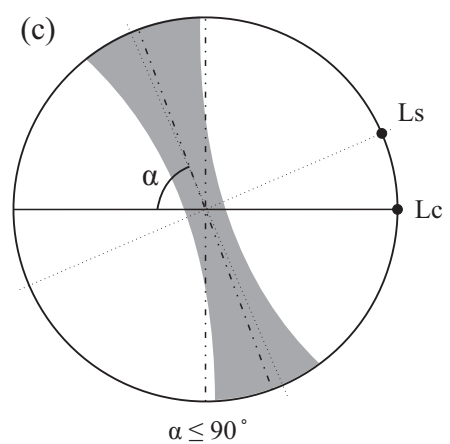
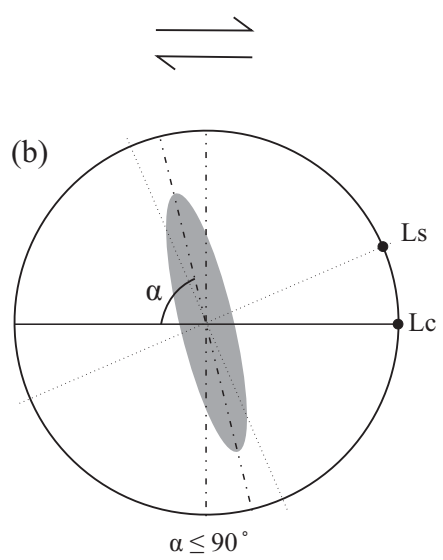
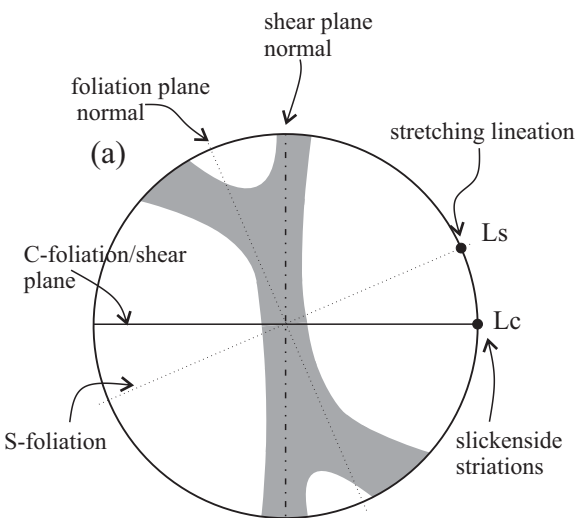
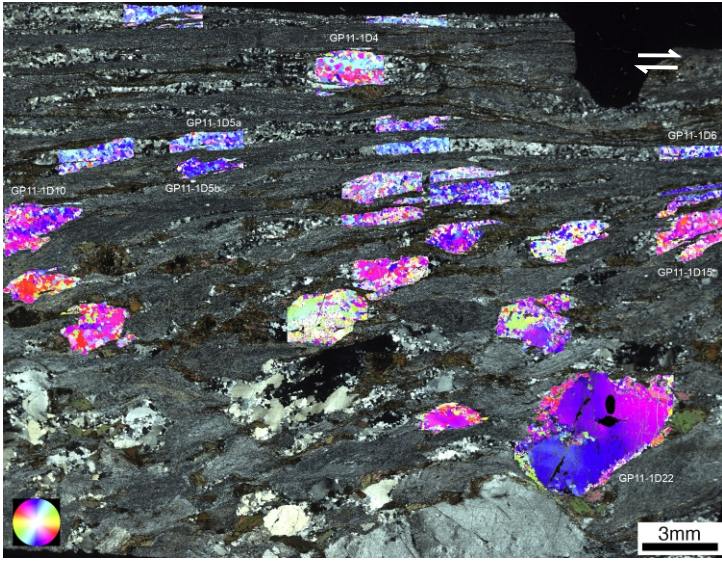
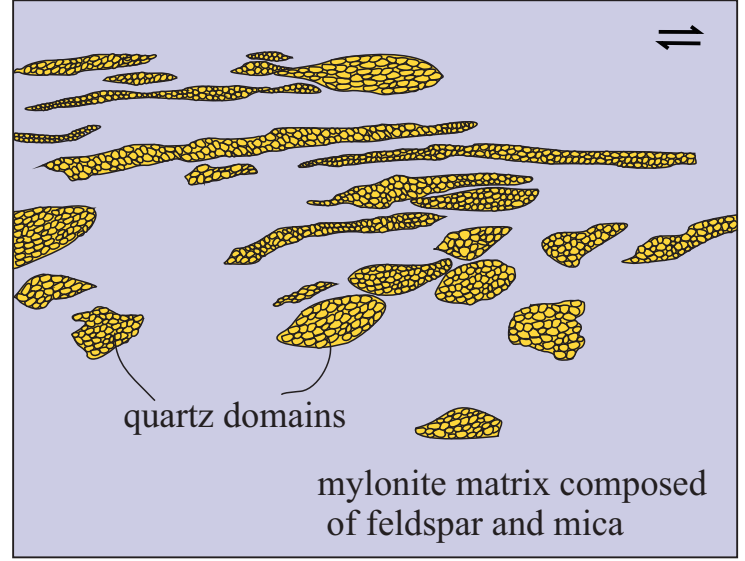


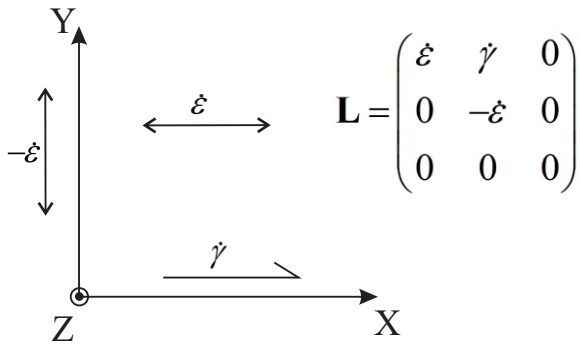
Figure 2.



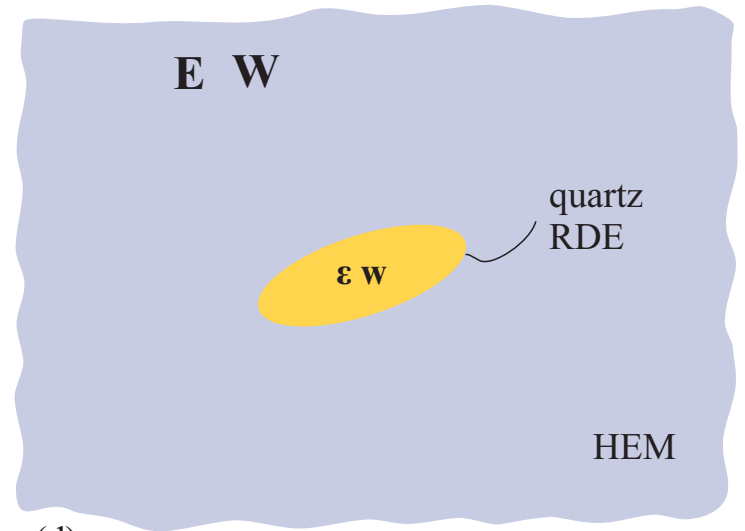
(a)



(b)



(c)



(d)

$$\epsilon = \mathbf{A}:\mathbf{E}$$

$$\mathbf{w} = \mathbf{W} + \mathbf{\Pi}:\mathbf{S}^{-1}:(\epsilon - \mathbf{E})$$

Figure 3.

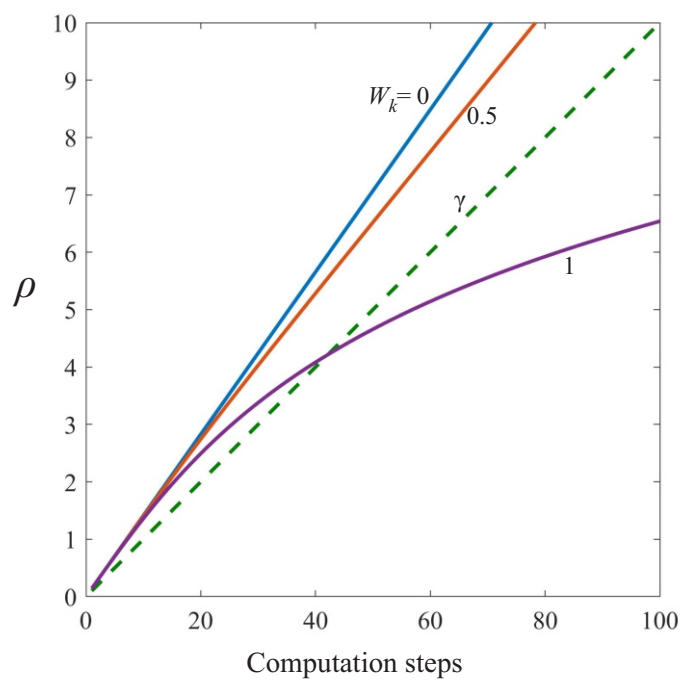
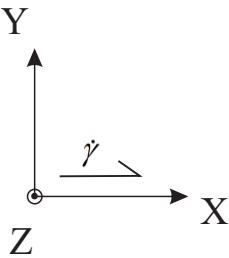
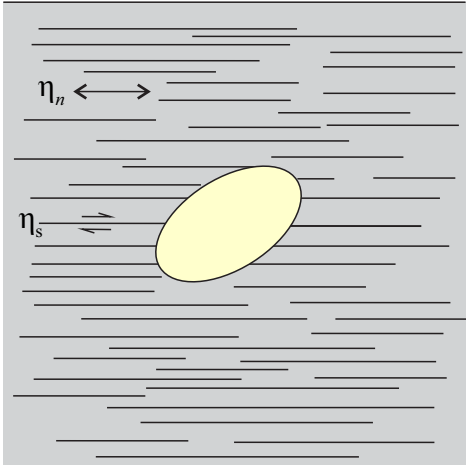




Figure 4.

$$m = \frac{\eta_n}{\eta_s}$$



$$\mathbf{L} = \begin{pmatrix} 0 & \dot{\gamma} & 0 \\ 0 & 0 & 0 \\ 0 & 0 & 0 \end{pmatrix}$$

Figure 5.

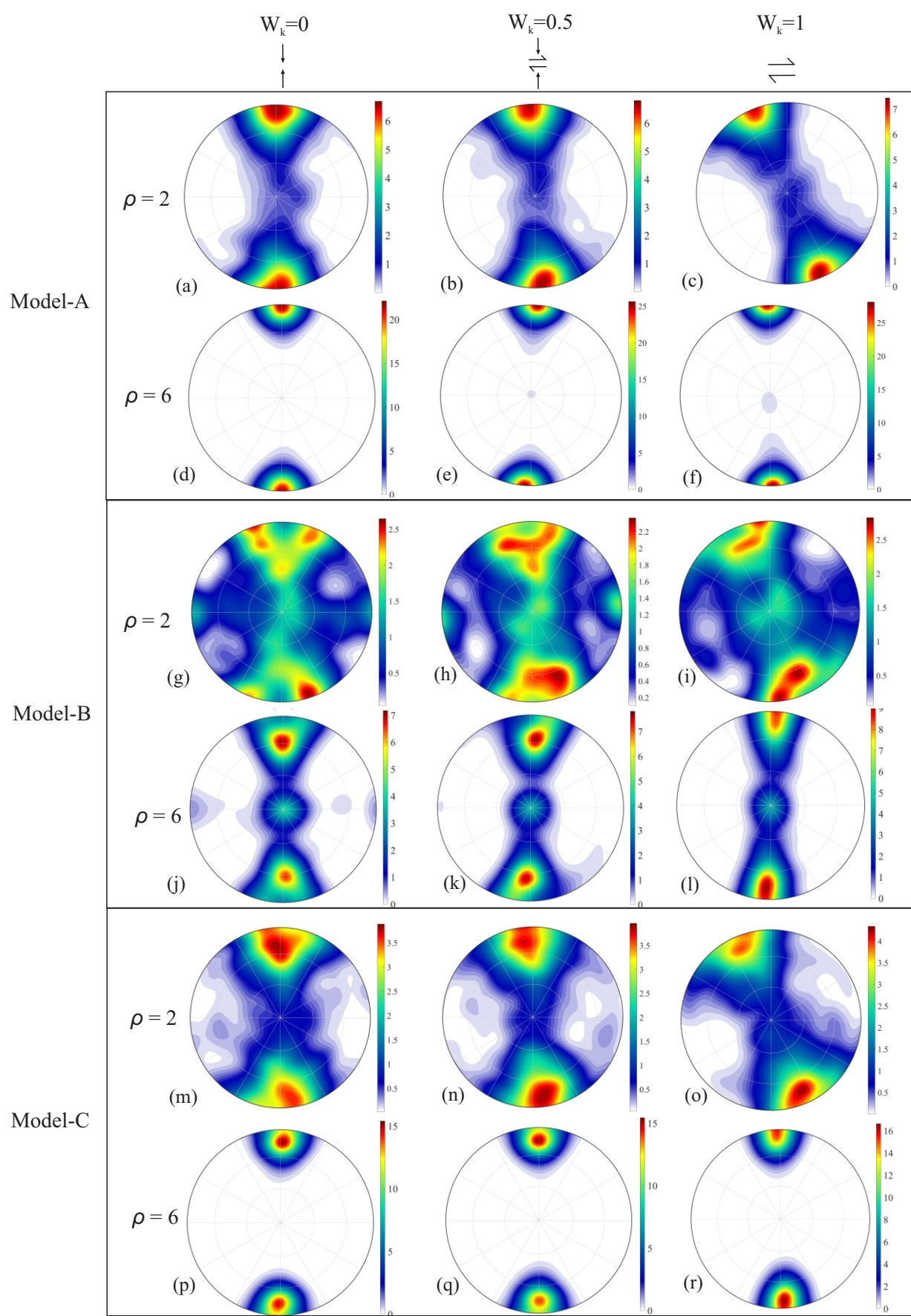


Figure 6.

$W_k=0$   
 $\downarrow$   
 $\uparrow$

$W_k=0.5$   
 $\downarrow$   
 $\uparrow$

$W_k=1$   
 $\rightleftharpoons$

Model-D

$\rho = 2$

(a)

(b)

(c)

$\rho = 6$

(d)

(e)

(f)

Model-E

$\rho = 2$

(g)

(h)

(i)

$\rho = 6$

(j)

(k)

(l)

Model-F

$\rho = 2$

(m)

(n)

(o)

$\rho = 6$

(p)

(q)

(r)

**Figure 7.**

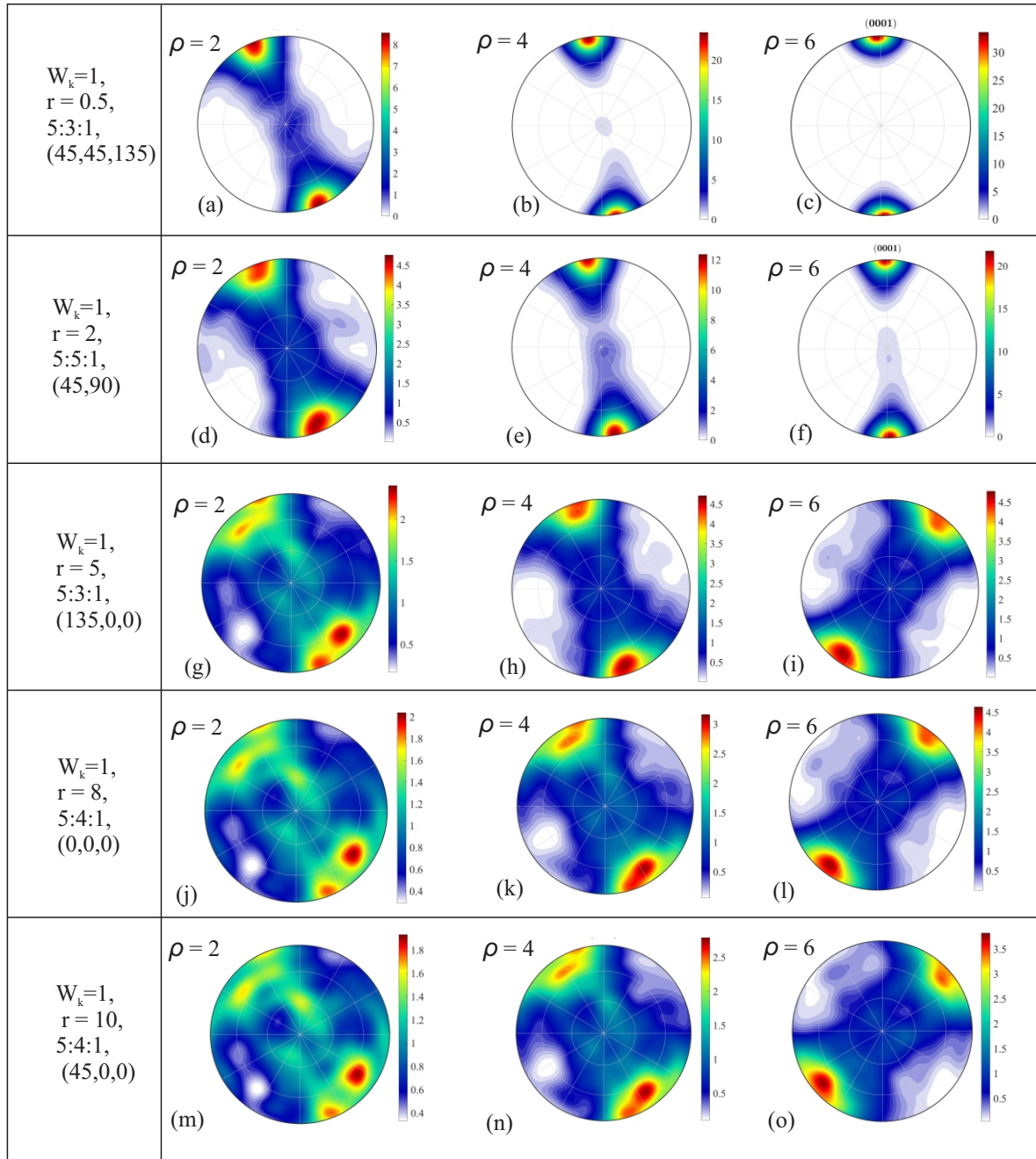




Figure 8.

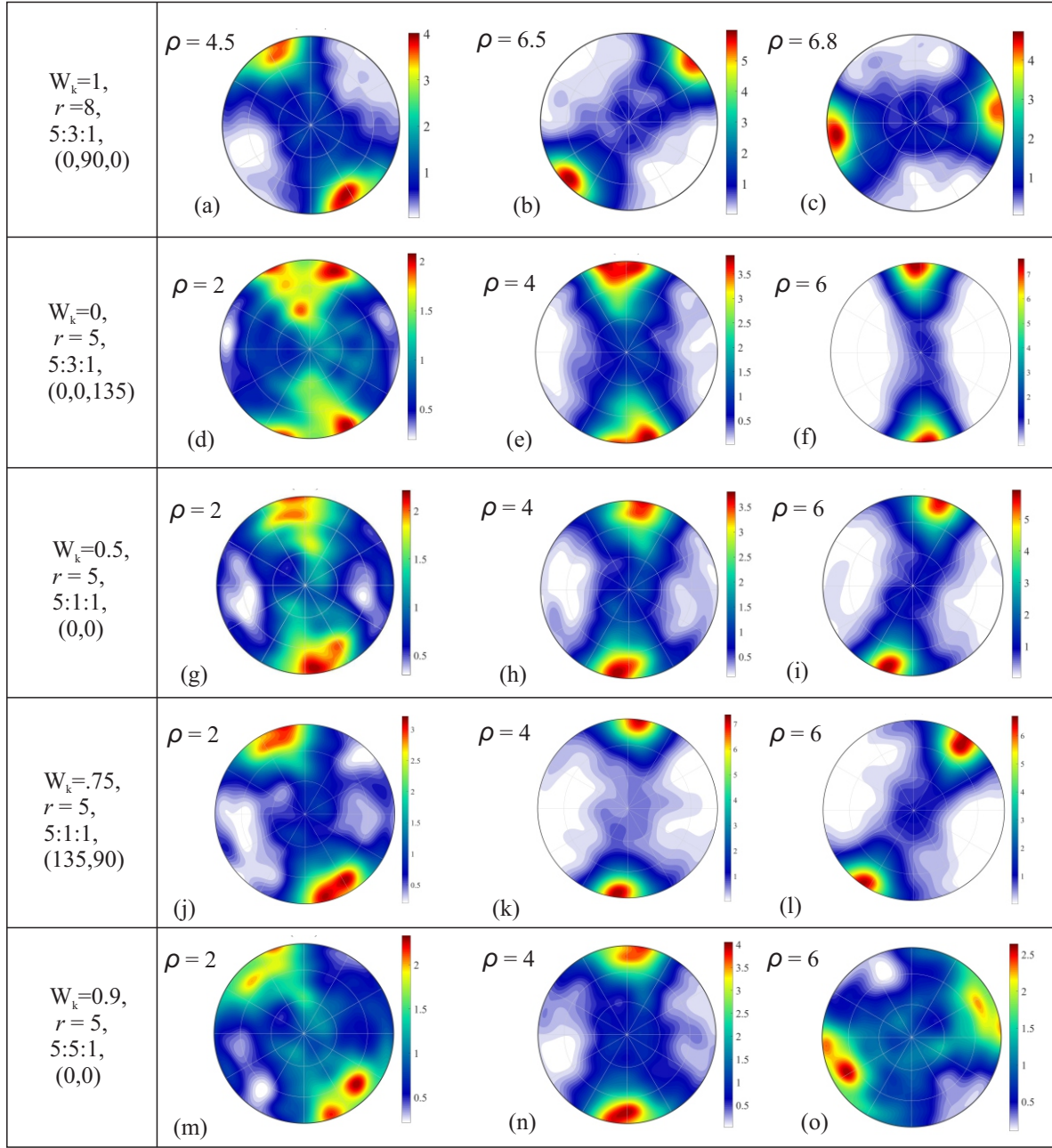


Figure 9.

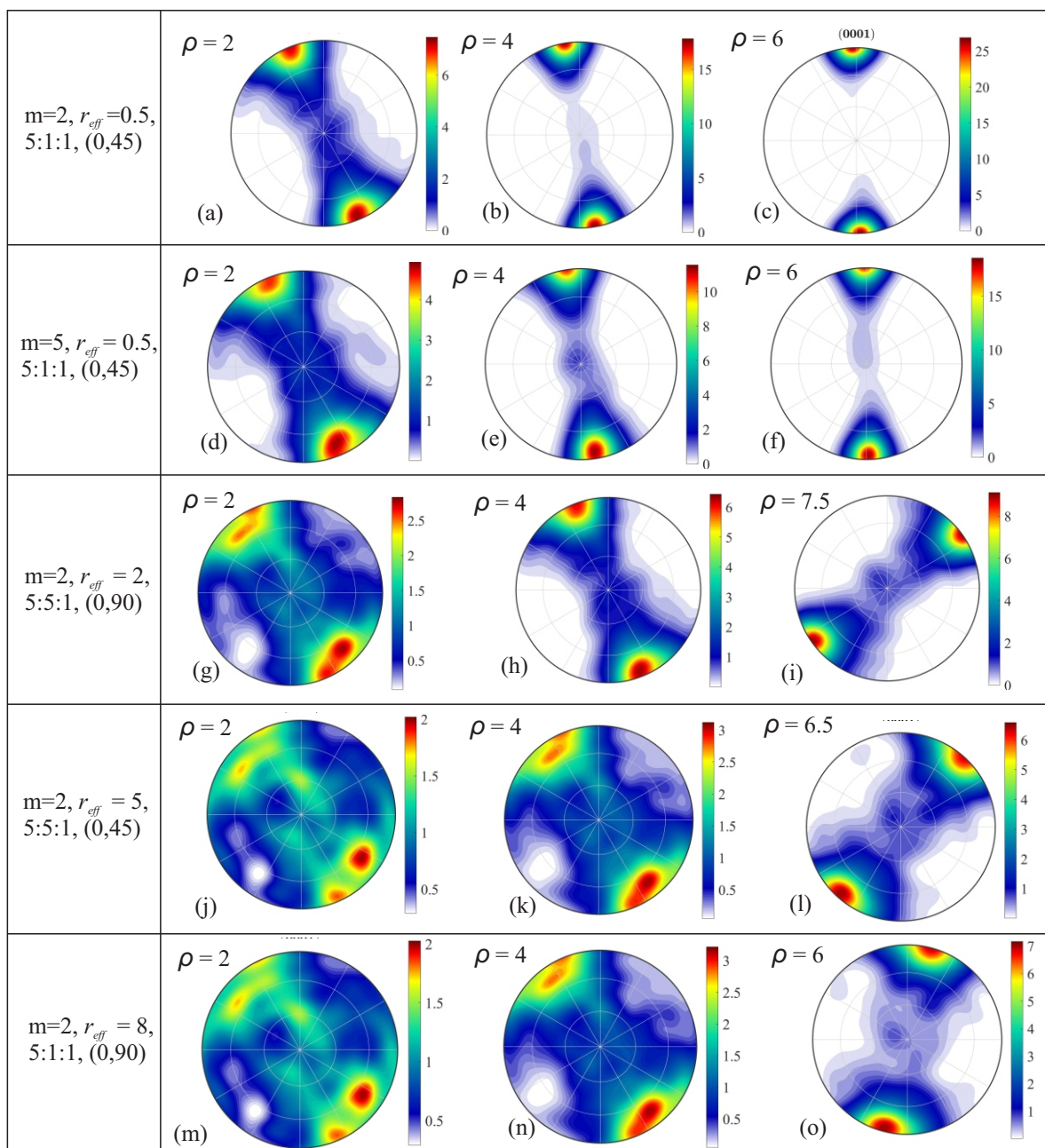


Figure 10.

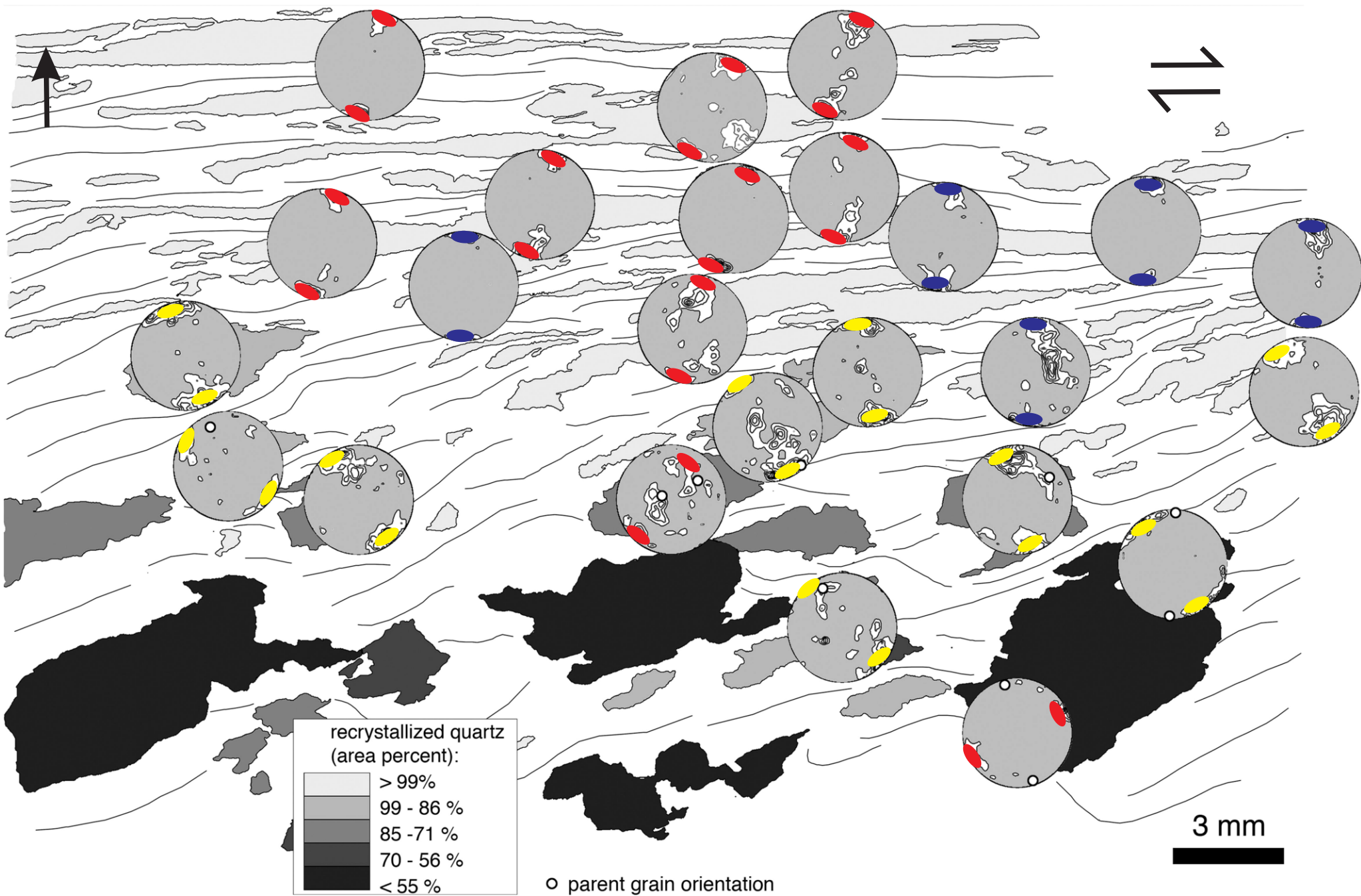
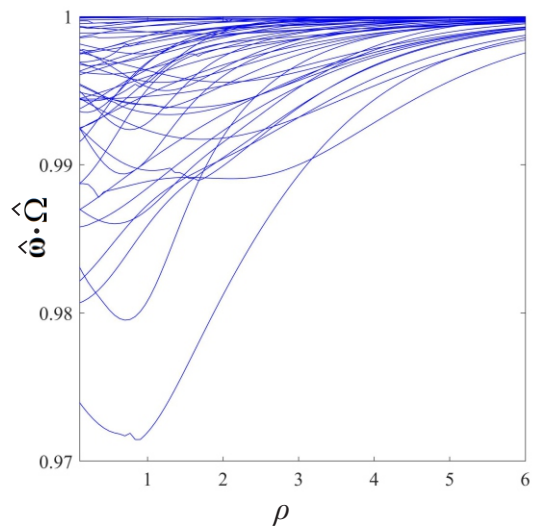
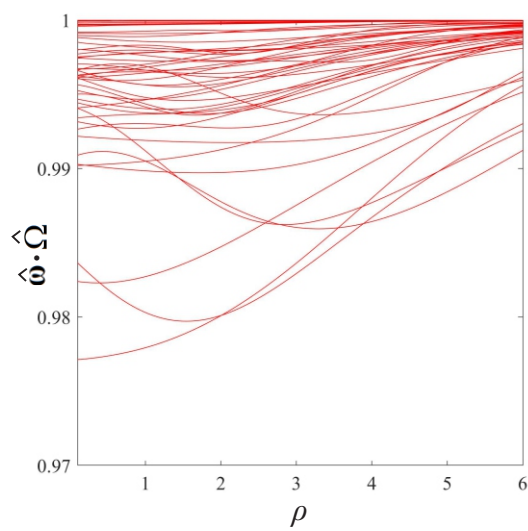
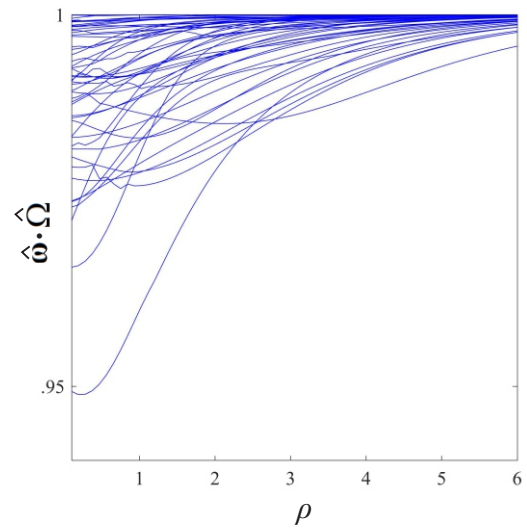
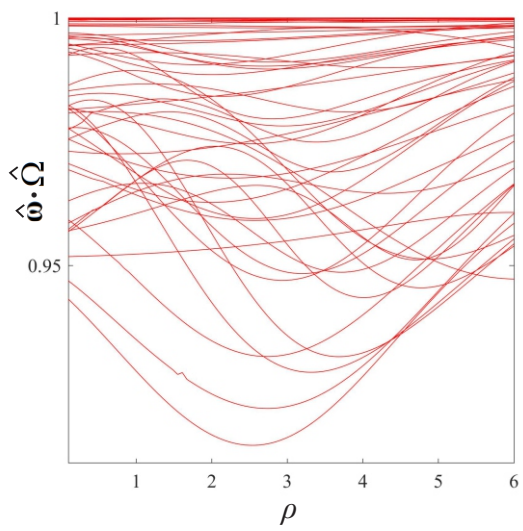
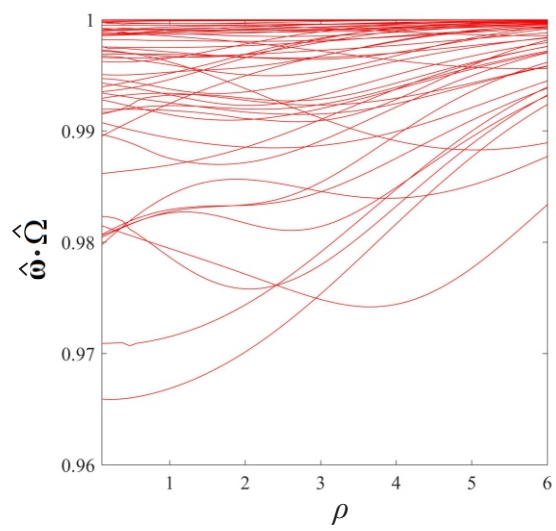
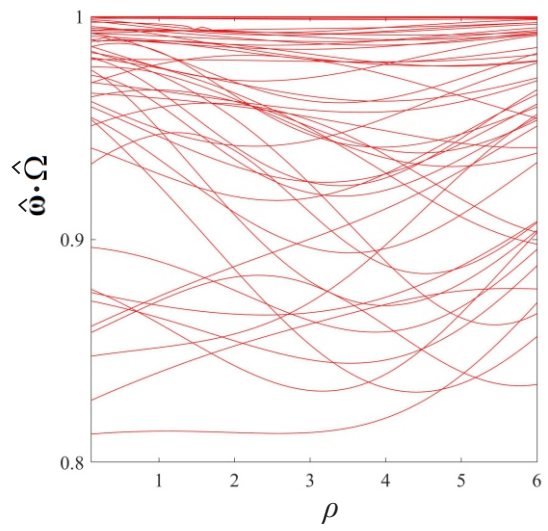


Figure 11.



$W_k=1$  $W_k=0.75$  $r=0.5$  $r=2$  $r=5$ 



Slip Systems	CRSS ratio					
	Model A	Model B	Model C	Model D	Model E	Model F
Basal <a> {0001} < $\bar{1}2\bar{1}0$ >	1	3	3	3	3	3
Rhomb<a> {10 $\bar{1}1$ } < $\bar{1}2\bar{1}0$ >	5	5	1	1	5	3
Prismatic<a> {10 $\bar{1}0$ } < $\bar{1}2\bar{1}0$ >	5	1	3	1	5	1
Prismatic <c> {10 $\bar{1}0$ } <0001>	10	10	10	10	1	1
Approximate Temperature Range	350-500°C	500-600 °C	500-600 °C	500-600 °C	650 °C	650°C