# Brittle Creep and Brittle Failure of Rocks: a reformulation of the wing crack model

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#### Abstract

We propose a reformulation of the wing crack model of brittle creep and brittle failure. Experimental studies suggest that the mechanical interactions of sliding and tensile wing cracks are complex, involving formation, growth and coalescence of multiple tensile, shear and mixed-mode cracks. Inspired by studies of failure in granular media, we propose that these complex mechanical interactions lead to the formation of micro shear-bands, which, in turn, develop longer wing cracks and interact with a wider volume of rock to produce larger shear bands. This process is assumed to indefinitely continue at greater scales. We assume the original wing crack formalism is applicable to micro shear-band formation, with the difference that the half-length, a, of the characteristic micro shear band is allowed to increase with deformation (i.e. wing crack growth). In this approach, the dimensionless shear band half-length A is related to the dimensionless wing crack length L by a function, A(L) = 1 + f(L), where f(L) embodies the entire process of shear band formation, growth and interaction with other shear bands and flaws and the problem is then to identify its proper form. We compare the model predictions for various classes of functions f(L)to experimental brittle creep data. Although a very large class of functions reproduce the classic sequence of tri-modal creep, we found that only the simple power law  $f(L) = (L/\Lambda)^q$  generated creep curves consistent with published creep data of rocks. Similar accord was also obtained with experimental brittle failure data.

#### 12 We propose a reformulation of the wing crack model of brittle creep and brittle failure. 13 Experimental studies suggest that the mechanical interactions of sliding and tensile wing cracks are 14 complex, involving formation, growth and coalescence of multiple tensile, shear and mixed-mode cracks. 15 Inspired by studies of failure in granular media, we propose that these complex mechanical interactions 16 lead to the formation of micro shear-bands, which, in turn, develop longer wing cracks and interact with a 17 wider volume of rock to produce larger shear bands. This process is assumed to indefinitely continue at 18 greater scales. We assume the original wing crack formalism is applicable to micro shear-band formation, 19 with the difference that the half-length, a, of the characteristic micro shear band is allowed to increase with 20 deformation (i.e. wing crack growth). In this approach, the dimensionless shear band half-length A is related to the dimensionless wing crack length L by a function, A(L) = 1 + f(L), where f(L) embodies the entire 21 22 process of shear band formation, growth and interaction with other shear bands and flaws and the problem 23 is then to identify its proper form. We compare the model predictions for various classes of functions f(L)24 to experimental brittle creep data. Although a very large class of functions reproduce the classic sequence 25 of tri-modal creep, we found that only the simple power law $f(L) = (L/\Lambda)^q$ generated creep curves consistent 26 with published creep data of rocks. Similar accord was also obtained with experimental brittle failure data. 27

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#### 29 Plain language summary:

31 Rocks close to Earth's surface deform by breaking. Breaking can occur abruptly if the load the rocks bear 32 increases rapidly. Breaking can however also occur over much longer times without changes to the load 33 during a process called brittle creep. Observations suggest that breaking occurs due to growth and linkage 34 of many small-scale flaws present in the rock. The details of this growth and linkage process, however, 35 are very complex which complicates our ability to assess when rocks will ultimately break. Here we 36 develop a model that simplifies the details of these small-scale interactions between large populations of 37 flaws into a simple functional form. We analyze a number of possible functional forms and find that the 38 simplest power law form yields good agreement with experimental data. Our model reproduces the 39 behavior observed in brittle creep experiments where, after a step increase in load, the initially rapid rate 40 of deformation first slows down, reaches a transitory steady-state and then accelerates until final failure 41 occurs. Our model hence improves our ability to predict when failure will occur and presents a step 42 towards mitigating the hazards associated with rock failure.

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# **Highlights:**

- Sliding cracks and their wing cracks interact with nearby flaws to form shear bands, which coalesce to form shear bands at greater scales.
- 48 Formation, growth and coalescence of shear bands is self-similar over a range of length-• 49 scales and is expressed by a power law function.

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• The model naturally reproduces trimodal creep curves and returns experimentally determinable quantities.

## 1. Introduction

55 Deformation in Earth's upper crust is dominated by fracturing and frictional sliding resulting in 56 macroscopically "brittle" behavior. Fractures occur over a range of length scales from intragranular microscopic cracks to fault zones spanning plate boundaries that host destructive earthquakes. The stress in 57 58 the upper crust is limited by frictional sliding on favorably oriented faults (e.g. Zoback and Zoback 2007) and therefore much focus was dedicated to the problem of sliding frictional interfaces which are the end-59 60 product of brittle failure (e.g. Marone 1998, Dietrich 2007). Brittle creep and brittle failure that precede the 61 formation of a through-going fault are relatively less studied phenomena, but nevertheless critical to our understanding of the long-term behavior of the crust and the earthquake cycle (e.g. Brantut et al. 2013, 62 63 Main 2000). These phenomena will be the focus of the present contribution.

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## 1.1 Background: the wing crack model

66 In the laboratory, rocks are typically tested under either constant stress or constant strain rate boundary conditions (or more precisely under constant load or constant displacement rate boundary 67 conditions). Typical stress-strain-time plots obtained for both types of tests are schematically illustrated in 68 69 figure 1. Under constant stress, rocks exhibit trimodal creep curves. Namely, decelerating "primary creep" 70 occurs after the initial change in stress, followed by a transitory, apparent steady state "secondary creep", 71 which eventually gives way to accelerating "tertiary creep" and failure (figure 1, left diagram). In constant 72 strain rate tests, rocks first deform elastically (albeit, often non-linearly) until the yield point (i.e., onset of 73 inelastic deformation), followed by strain hardening and accumulation of permanent strain. The axial stress 74 eventually reaches a maximum (peak stress or strength), at which point a fault starts developing and the 75 rock weakens to a stress level dictated by the residual friction on the fault (figure 1, right diagram). This 76 macroscopic behavior is controlled by the activation, propagation and interactions of cracks in the rocks in 77 the brittle regime. Loading conditions in nature are generally more complex than those employed in 78 experiments, nevertheless laboratory tests can provide valuable insights into the micromechanics of brittle 79 creep and brittle failure



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Figure 1. Schematic of typical curves of stress and strain versus time recorded in laboratory constant stress (left diagram) and constant strain rate tests (right diagram).

85 We are trying, here, to revisit the wing crack model of brittle failure and brittle creep. The wing 86 crack model attributes the rock inelastic deformation leading to failure to tensile cracks that emanate from 87 the tips of the largest pre-existing microcracks undergoing frictional shear. These activated microcracks are 88 called the dominant cracks and are represented in the model by a single characteristic crack length,  $2a_0$ , and

89 their inclinations with respect to the remotely applied principal stresses,  $\sigma_3 < \sigma_2 < \sigma_1$ . The resolved shear

90 stress is maximum in cracks oriented parallel to the intermediate stress  $\sigma_2$  and inclined with respect to the 91 minimum stress  $\sigma_3$  by an angle  $\beta = 45^\circ$  (figure 2). The model assumes that the dominant cracks verify these 92 optimal conditions, implying that the intermediate stress plays no role in the process of rock failure except 93 for controlling the orientation of the final fracture. We will not attempt to modify this assumption in our 94 model. A discussion of the effect of  $\sigma_2$  is considered out of the scope of the present paper. 95



 $\int \sigma_1$ 

Figure 2. Schematic of a sliding microcrack and the associated wing cracks. The intermediate stress  $\sigma_2$  is normal to the figure plane. The angle  $\beta$  is, hereafter, assumed to be 45°.

100 In the idealized conditions depicted in figure 2, the wing cracks are curved. The angle  $\theta$  they form with the 101 dominant microcrack changes during propagation, starting at about 70° at initiation and decreasing until 102 the wing cracks become parallel to  $\sigma_1$ . Assuming  $\beta = 45^\circ$ , the normal stress and the resolved shear stress 103 on the dominant microcrack are:

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	$\sigma_{\rm N} = \frac{1}{2} \left( \sigma_1 + \sigma_3 \right)$		(1)
	$ \tau  = \frac{1}{2} (\sigma_1 - \sigma_3)$		(2)
		00 1 1	1

106 Part of the shear stress is balanced by friction. The effectively active shear stress is therefore given by:

107  $|\tau_{eff}| = |\tau| - \mu \sigma_N$  (3) 108 where  $\mu$  is the friction coefficient and the condition  $|\tau| \ge \mu \sigma_N$  imposes  $\sigma_1/\sigma_3 \ge (1-\mu)/(1+\mu)$ . Note that we 109 are using the geophysics convention that compressive stresses are positive.

110 The mode I stress intensity factor,  $\kappa_{I}$ , of a wing crack is a complex function of the remotely applied 111 principal stresses, the length  $2a_{0}$  of the dominant crack, the wing crack length *l*, and, the angles  $\beta$  and  $\theta$ 112 (figure 2). A number of models have been published (see Baud et al., 1996, for a review). Although they 113 differ in some details depending on how the curvature of the wing cracks and other such features are treated, 114 they all consist of the sum of two terms, one driving and one resisting wing crack propagation. Here, we 115 consider the simple model from Kachanov (1982) (also used in Brantut et al., 2013):

$$\kappa_{\rm I} = -1.15 |\tau_{\rm eff}| \sqrt{\pi a_0} + \sigma_3 \sqrt{\pi l/2} \tag{4}$$

117 Despite its simplicity (e.g., the wing crack curvature is ignored), the Kachanov model considers 3D penny 118 shape microcracks and not 2D cracks like many other models. The model predicts that wing cracks will 119 form when the initial stress intensity factor is larger than the rock fracture toughness,  $|\kappa_0| =$  $1.15 |\tau_{eff}| \sqrt{\pi a_0} \ge |\kappa_{Ic}|$ , or, in other words, when  $|\tau_{eff}|$  exceeds a critical shear stress  $\tau_c = 1.15 |\tau_{eff}| \sqrt{\pi a_0} \ge |\kappa_{Ic}|$ 120  $|\kappa_{\rm Ic}|/(1.15\sqrt{\pi a_0})$ . The first (driving) term in equation 4 expresses the wedging effect of shear 121 122 displacements at the tips of the dominant crack. The second (resistant) term depends on l and accounts for 123 wing crack closure caused by  $\sigma_3$ . The two terms have different signs, causing  $|\kappa_1|$  to decrease with increasing 124 *l*. Wing crack growth will therefore stop when  $\kappa_I$  becomes equal to  $\kappa_{Ic}$ , or, in other words, when the wing

crack length reaches  $l_0 = a_0 (1.15 (|\tau_{eff}|-\tau_c)/\sigma_3)^2$ . This ultimate crack arrest means that the behavior of a 125 single wing crack system cannot be used to model brittle failure except, perhaps, in uniaxial compression 126 conditions (increasing  $\sigma_1$  and  $\sigma_3 = \sigma_2 = 0$ ), when the wing cracks eventually intersect the sample edges (i.e., 127 axial splitting). But this difficulty can be resolved by recognizing that the rock contains a broad population 128 129 of mechanically interacting flaws. Interacting dominant cracks are expected to experience an increase of  $|\kappa_1|$  during wing crack extension. The currently published crack interaction models are all variations of the 130 131 model developed by Ashby and Hallam (1986) and Ashby and Sammis (1990). In this model, the dominant microcracks are assumed to form linear arrays parallel to the maximum stress  $\sigma_1$ . Their growing wing cracks 132 133 are therefore colinear and, owing to their mutual influence, the remotely applied lateral stress  $\sigma_3$  is locally 134 reduced in the ligaments between neighboring wing cracks by a quantity  $\sigma_i$ . This causes a decrease of the resistant term,  $(\sigma_3 - \sigma_i)\sqrt{\pi l/2}$ , and allows further propagation and finally coalescence of the wing cracks. The wing cracks thus form columns parallel to the maximum stress  $\sigma_1$ , which ultimately fail, owing to the 135 136 137 classic buckling instability of slender columns. The intrinsic weakness of this model is the assumption of a 138 very specific geometrical structure of the dominant microcracks, which is very unlikely to be found in a 139 natural material. Nevertheless, the model has been quite successfully applied to experimental rock 140 deformation and failure data (see the review by Brantut et al., 2013).

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#### **1.2 Background: crack coalescence**

143 Since the early work of Horii and Nemat-Nasser (1985), the interaction and coalescence of wing 144 cracks in conditions of uniaxial and biaxial compression have been experimentally investigated in a variety 145 of materials, including rocks (see the comprehensive review of Wong, 2008). One important result is that 146 biaxially loaded samples containing two parallel, cm-scale man-made crack-like flaws produced strongly 147 different coalescence patterns depending on their relative positions and inclinations with respect to the 148 applied principal stresses (e.g., Wong and Einstein, 2009ab; Lin et al., 2021). In Lin et al. (2021), a scalar 149 measure of the strain field was determined as a function of time using digital image correlation analysis, 150 allowing identification of the fissures developing and coalescing around the initial flaws. In the configuration shown in figure 3 (a schematic partial reproduction of Lin et al.'s figure 12), the wing cracks 151 152 emanating from the inclined flaws merged and two new ones were created on the sides, forming a column 153 aligned with  $\sigma_1$  quite similar to the structure assumed in the Ashby-Hallam-Sammis model. However, the 154 unstable buckling of the column predicted in the Ashby-Hallam-Sammis model was not observed, perhaps 155 because the experiment was prematurely stopped.

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33.3 MPa



163 In other configurations of the initial flaws, very different coalescence patterns occurred, in which the

inclined shear cracks themselves were actively involved. For example, a separate, small shear crack formed

between two nearly aligned initial flaws and eventually merged with both of them, completely bridging the

ligament (figure 4). Very complex bridging structures combining tensile, shear and mixed mode crackswere also observed in other configurations of the initial flaws.

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169 19.8 MPa 25.3 MPa 32.2 MPa
 170 Figure 4. Simplified reproduction of figure 7 of Lin et al. (2021). Here the position of the initial flaws leads to the formation of a shear crack and the eventual bridging of the ligament between the initial flaws.

Similar coalescence patterns were produced in samples of Carrara marble and molded gypsum (Wong and
Einstein, 2009ab). For example, nearly aligned flaws produced bridging shear cracks, while merging of the
tensile wing cracks occurred when the flaws were shifted to form a 90° angle. Combinations of shear, tensile
and mixed mode bridging cracks were observed in other cases.

# 2. The model

# 2.1 The concept: micro to macroscopic shear bands

180 Despite the general similarity of the results described above, important differences were noted in 181 materials with distinct internal structures. For example, development of a secondary crack was usually 182 preceded in Carrara marble by an increase in light reflectivity (whitening) of the sample surface in a thin region exactly delineating the path of the future secondary crack (Wong, 2008; Wong and Einstein, 183 184 2009ab). These "white patches" visible on the sample surface prior to cracking can be attributed to the 185 formation and accumulation of damage in highly strained thin zones. This explanation implies the existence, 186 in the material, of a population of very small defects and microcracks (invisible to the eye or even a very 187 high-resolution camera) that are activated by the amplified stresses around the large cm-scale flaws. White patches were not observed in molded gypsum, a microgranular material that was fabricated using 188 189 procedures specifically designed to ensure excellent homogeneity. Lajtai (1974) similarly noticed the 190 formation and growth of damaged shear zones at the tips of a sheared cm-scale synthetic crack in a biaxially 191 stressed plaster slab. Evidence of interaction of large propagating cracks with smaller flaws is also reported 192 in Brantut et al. (2014a). Their figure 12 shows examples of, on one hand, very smooth and rectilinear wing 193 cracks in homogeneous sparitic calcite cement and, on the other, rough and tortuous cracks traversing 194 microgranular micritic aggregations. As in rocks, failure of unconsolidated granular aggregates is generally 195 observed to entail strain localization on shear bands. The initiation and growth of shear bands in granular materials has been extensively investigated experimentally and numerically (e.g., see Desrues, 1990, and 196 197 references therein). One important observation is that the first stage of strain localization consists in the 198 formation of multiple, separate micro shear bands. With increasing stress, some of these micro shear bands 199 interact with neighboring ones, coalesce and eventually develop into macroscopic shear bands (Desrues 200 and Andó, 2015). Similar scenarios have been observed in rocks. For example, figure 16 in Brantut et al. 201 (2013) shows the hypocenter locations of acoustic emissions recorded during a creep test of a granite

sample. The distribution of hypocenters is random and featureless during primary and secondary creep but
 becomes strongly concentrated during tertiary creep in a region largely coincident with the final shear
 fracture (see also, Lockner et al., 1992; Lockner, 1993; Fortin et al., 2009; Fortin et al., 2010).

205 Here, we follow a similar concept. Natural flaws and microcracks in rocks have a very broad 206 distribution of sizes. The (largest and most favorably oriented) dominant cracks are the first to develop wing cracks, which, as discussed above, are bound to interact with nearby minor flaws. We posit that this 207 208 process will generally result in the development, near the tip of the dominant crack, of a complex array of 209 microcracks (formed in tensile, shear and mixed mode) that globally deforms in shear. For lack of a better 210 term, we will use the granular material terminology and call these structures micro shear bands. We 211 emphasize that this term is meant to cover a broad range of damage structures, from simple merged shear 212 cracks to inclined zones of crushed material. All these structures share a greater susceptibility to deform in 213 shear than their surrounding, hence our choice to refer to them as shear bands. Shear cracks and micro shear 214 bands similarly slide when submitted to sufficient shear stresses, although they may have different effective 215 coefficients of friction. A wing crack should therefore develop at the outer edge of the shear band (figure 216 5). Note that wing cracks have indeed been observed to initiate from the edges of sheared microstructural 217 objects other than microcracks. For example, Rawling et al. (2002) present SEM images of wing cracks 218 emanating from the edges of sheared biotite grains in triaxially deformed samples of Four-mile gneiss (e.g., 219 see their figure 9). Furthermore, owing to the overall increase in length, the wing crack emanating from a 220 micro shear band will be longer than the initial one and, thus, have the capacity to interact with a greater 221 volume of rock. It becomes therefore likely that two micro shear bands in favorable positions and 222 orientations will coalesce to form even longer micro shear bands, from the edges of which increasingly long 223 wing cracks will grow (figure 5). The model does not include an upper cut-off scale and the process is 224 therefore assumed to continue at indefinitely increasing scales.

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Figure 5. Schematic representation of mechanical interactions of cracks and flaws and the resulting micro shear band formation. The change of scale shown here is assumed to continuously take place at greater and greater scales.

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#### 2.2 Micro shear band model: brittle failure

We posit that the brittle deformation of rock illustrated in figure 5 can be described as the result of crack growth and coalescence around a representative dominant (or leading) micro shear band. Given the mechanical similarity of shear bands and sheared cracks, it is reasonable to assume that a leading micro shear band can be also modeled using the mathematical framework expressed in equation 4, the only difference being that the half-length *a* of the leading micro shear band increases during loading.

Since the second term of equation 4 describes the restraining effect of  $\sigma_3$  on wing crack expansion, it does not need to be modified. In the first term, the friction coefficient entering the definition of  $\tau_{eff}$  may depend on the internal structure of the shear band and thus vary during crack growth. For the sake of simplicity, we will assume that the friction coefficient of the micro shear bands remains approximately equal to that of the sheared cracks. Equation 4 then becomes:

$$\kappa_{\rm I} = -1.15 |\tau_{\rm eff}| \sqrt{\pi a(l)} + \sigma_3 \sqrt{\pi l/2} \tag{5}$$

- 242 where the half-length of the micro shear band is an increasing function of l. If we assume that wing crack 243 growth is stable (as it has to be when the mechanical interactions are negligible) the crack arrest condition,
- 244  $\kappa_{I} = \kappa_{Ic}$ , yields:

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$$-1.15(|\tau_{\rm eff}| - \tau_{\rm c})\sqrt{\pi a(l)} + \sigma_3\sqrt{\pi l/2} = 0 \tag{6}$$

If  $|\tau_{eff}|$  is considered the independent variable (i.e., in constant loading rate tests), solving equation 6 for a(l)246 247 yields:

$$a(l) = l(\sigma_2/1.15(|\tau_{\rm eff}| - \tau_{\rm c}))^2$$

which simply states that a(l) can be calculated if the dependence of l on  $|\tau_{eff}|$  is known. In this description, 249 failure occurs with increasing  $|\tau_{eff}|$  when the leading shear band eventually reaches the sample boundaries. 250 251 Rock failure tests, however, are usually carried out in constant strain rate conditions and not at constant 252 loading rate as assumed by equation 7. In this case, l is the independent variable and prior knowledge of 253 a(l) is needed to model brittle failure. 254

#### 2.3 Micro shear band model: brittle creep

256 Following previous models of brittle creep, we introduce time dependence by assuming subcritical 257 crack growth. In this case, wing crack propagation can proceed at constant stresses below the critical shear stress  $\tau_c$ . The initial stress intensity factor is  $|\kappa_0| = 1.15 |\tau_{eff}| \sqrt{\pi a_0} < |\kappa_{Ic}|$  and equation 5 can be recast as: 258

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$$\frac{\kappa_{\rm I}}{\kappa_0} = \sqrt{a(l)/a_0} + \frac{\sigma_3}{\kappa_0}\sqrt{\pi l/2} \tag{8}$$

Equation 8 can be further simplified by using the following dimensionless variables and parameters:  $K_I =$ 260  $\kappa_{\rm I}/\kappa_0$ ,  $A = a/a_0$  and  $L = l/l_0$ , where  $l_0 = 2a_0 (1.15 |\tau_{\rm eff}|/\sigma_3)^2 = 2 \kappa_0^2/(\pi \sigma_3^2)$ , i.e., the maximum possible length 261 262 of the wing cracks generated under the current state of stress at the tips of a microcrack of length  $2a_0$  in the absence of any mechanical interactions ( $l_0$  is thus an intrinsic property of the dominant microcracks of the 263 264 undeformed rock). Note that  $l_0$  can be physically achieved when subcritical crack growth is operating since 265  $|\kappa_I|$  is allowed to drop to zero. We finally obtain:

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(9)

(7)

 $\mathbf{K}_{\mathrm{I}} = \sqrt{A(L)} - \sqrt{L}$ Although equation 9 is not a mechanistic model of crack coalescence, the function A(L) is effectively a 267 closed-form expression of the results of the extremely complex and varied mechanical interactions 268 269 underlying the formation and growth of micro shear bands. To do the job correctly, A(L) must satisfy a 270 number of constraints. First, there are the trivial conditions that A(0) = 1 and A(L) must be a monotonically 271 increasing function. Thus, A(L) can be expressed as A(L) = 1 + f(L), where f(L) is a monotonically increasing 272 function verifying f(0) = 0. Most importantly, K<sub>1</sub> must always be strictly larger than zero. Negative values 273 obviously contradict the definition of K<sub>1</sub> since  $\kappa_1$  and  $\kappa_0$  are both negative quantities. Moreover, K<sub>1</sub> = 0 274 implies crack arrest and is therefore incompatible with tertiary creep.

275 The simplest functions satisfying these conditions are the power law functions,  $f(L) = (L/\Lambda)^q$ , where 276  $\Lambda = \lambda/l_0$  is a dimensionless length scale representing the normalized distance that the wing crack must 277 propagate to enter the vicinity of another microcrack or flaw, interact with it and therefore produce a substantial increase of K<sub>I</sub>. Of course, other more complex functions such as the polynomials, f(L) =278  $\sum_{i=1}^{q} C_i L^i$  (with  $C_i \ge 0$ ), can also meet the conditions above. We will focus here on the power laws f(L) =279  $(L/\Lambda)^q$ , because they are the analytically simplest functions and, as will be discussed later, produced results 280 281 in good agreement with experimental brittle creep data. We nevertheless investigated the polynomials f(L)282  $= (1 + L/\Lambda)^q$  - 1 (i.e.,  $A(L) = (1 + L/\Lambda)^q$ ) as thoroughly as  $f(L) = (L/\Lambda)^q$  but the results obtained were 283 inconsistent with experimental brittle creep observations and will not be discussed in detail in the following 284 text.

To illustrate how equation 9 works, it is convenient to consider the perfect square function A(L) =285  $(1+L/\Lambda)^2$  (see figure 6). Since the square root of A(L) is a linear function of L, the first term in equation 9 286 287 yields a family of straight lines with slopes increasing when  $\Lambda$  is decreased (the values of  $\Lambda$  are indicated 288 in the same color as the corresponding lines in figure 6). The values of  $K_I$  are graphically measured as the 289 vertical distance between these lines and the  $\sqrt{L}$  curve representing the second term of equation 9 (dashed

290 line in figure 6). Thus, in all cases, K<sub>I</sub> first decreases (primary creep), reaches a minimum (secondary creep)

and then increases indefinitely (tertiary creep). To satisfy the condition,  $K_I > 0$ , the colored straight lines in

figure 6 are not allowed to intersect or tangentially touch the  $\sqrt{L}$  curve, thus limiting the values that  $\Lambda$  can

take (in this example,  $\Lambda < \Lambda_c = 4$ ). We also note that increasing  $\Lambda$  causes a reduction of the minimum K<sub>m</sub> of K<sub>I</sub> and of the rate of increase of K<sub>I</sub> in the second stage while the value  $L_m$  of L at the minimum increases (figure 6). K<sub>m</sub> and  $L_m$  are key output parameters of the model since they identify the transition from primary to tertiary creep.

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298 *L* 299 Figure 6. Graphic representation of equation 9 in the case of  $A(L) = (1+L/\Lambda)^2$ . The vertical distance between the colored 300 straight lines (the values of  $\Lambda$  for each line are indicated in matching colors) and the dashed purple curve is a measure of the stress 301 intensity factor K<sub>1</sub>. The position  $L_m$  of the minimum stress intensity factor K<sub>m</sub> of the black curve is indicated by the small empty 302 circle and the associated vertical dashed line. Note that K<sub>m</sub> is nearly equal to zero in this case. See text for a detailed interpretation. 303

The consecutive K<sub>I</sub>-decreasing and K<sub>I</sub>-increasing stages illustrated in figure 6 were produced by the power laws  $f(L) = (L/\Lambda)^q$  with an exponent  $q \ge 2$  as well as all other functions f(L) examined in this study. These various f(L) functions, however, differed in the values of  $\Lambda_c$ , K<sub>m</sub> and L<sub>m</sub> that they generated.

307 To assess the effect of the power law exponent *q* we examined the form of the curves of K<sub>I</sub> versus 308 *L* associated with  $f(L) = (L/\Lambda)^q$  for different values of *q* and  $\Lambda$  (the curves associated to q = 2 to 6 and  $\Lambda =$ 309 1 are shown in figure 7).



Figure 7. Examples of curves of stress intensity factor K<sub>I</sub> versus wing crack length L associated with  $f(L) = (L/\Lambda)^q$  with  $\Lambda = 1$  for various values of the exponent q as indicated in matching colors. See text for a detailed discussion.

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The limit  $\Lambda_c$  is analytically related to the exponent q through the expression,  $\Lambda_c = (q^{1/(1-q)} - q^{q/(1-q)})^{(1-q)/q}$  (see 315 316 Appendix A for the derivation). According to this relation,  $\Lambda_c$  decreases from 2 to ~1.46 when q is increased from 2 to 8. As mentioned earlier,  $\Lambda = \lambda/l_0$  represents the normalized distance, over which a wing crack 317 318 must propagate to enter the vicinity of other microcracks and flaws and interact with them. It is important 319 to note that this interaction distance depends both on the properties of the material being modeled and the 320 state of stress considered. Indeed, the effect of the stress state is introduced through the normalization factor  $l_0 \propto (|\tau_{\text{eff}}|/\sigma_3)^2$  while the material is described by both  $l_0$ , which contains  $a_0$ , and  $\lambda$ , which can be understood 321 as a characteristic flaw separation (or the inverse of the flaw density). Consequently, variations of  $\Lambda$  may 322 323 represent either distinct materials with differing flaw densities or a single material subjected to various levels of stress. Thus, an increase of q indicates an increase of either the flaw density of the rock considered 324 325 or the level of stress needed to allow brittle creep. We also note that the minimum stress intensity factor K<sub>m</sub> 326 along the K<sub>I</sub> versus L curves decreased with increasing q while the corresponding  $L_m$  first decreased and 327 then increased (figure 7).

328 Time dependence is then introduced in the model by assuming subcritical crack growth. According 329 to this assumption, crack propagation proceeds gradually, starting at a stress intensity factor  $\kappa_0$  lower than 330  $\kappa_{IC}$ . Although different sub-critical crack growth models can be used, the most commonly reported in 331 previous studies is based on the power law relation often called Charles' law (Charles, 1958; see also 332 Wiederhorn and Boltz, 1970):

 $\frac{v}{v_0} = \left(\frac{\kappa_I}{\kappa_0}\right)^n$ 333 (10)

where v and  $v_0$  are propagation velocities of the wing cracks, and the exponent n usually takes very high 334 values (between 10 and 50). Note that equation 10 is dimensionless and can be re-written  $V = K_{I}^{n}$ , with V =335 336  $v/v_0$ , and then combined with equation 9 to yield:

337 
$$V = \frac{dL}{dT} = \left(\sqrt{1 + f(L)} - \sqrt{L}\right)^n \tag{11}$$

where the normalized time is defined as  $T = t/t_0 = t v_0/l_0$ . Using the previously discussed functions  $f(L) = t v_0/l_0$ . 338  $(L/\Lambda)^q$ , equation 11 contains three parameters, the normalized flaw separation  $\Lambda$ , the power law exponent 339 q and the subcritical crack growth exponent n. Since L is a strictly monotonic, increasing function of T, 340 341 equation 11 can be numerically solved using the following simple procedure. First we construct a wing crack length series,  $L_i = (i - 1) dL$ , where dL denotes a small increment (e.g., dL = 0.01 or lower for more 342 343 accuracy). We then use equation 10 to calculate the corresponding wing crack velocity series  $V_i$ . Finally, the time series  $T_i$  can be calculated using the recurrence  $T_{i+1} = T_i + dT_i$ , with  $dT_i$  calculated by numerical integration of  $dT_i = \int_{L_i}^{L_{i+1}} (\sqrt{1 + f(L)} - \sqrt{L})^{-n} dL$ . Examples of curves of *V* as a function of *T* corresponding 344 345 to  $f(L) = (L/\Lambda)^q$  with  $\Lambda = 0.1, 0.5, 1., 1.4$  and 1.75, q = 2 and 4, and, n = 10 are shown in figure 8. 346 347





Figure 8. Examples of curves of wing crack growth rate V versus time T associated with  $f(L) = (L/\Lambda)^q$  for q = 2 (left diagram) and 4 (right diagram). The values of  $\Lambda$  corresponding to each curve are indicated in matching colors (the subcritical crack 351 growth exponent n was equal to 10 in all cases). The times  $T_{inf}$ ,  $T_2$  and  $T_F$ , and the exponent m are graphically defined in the inset. 352 See text for a detailed discussion.

353

In all cases considered in this study, the *V* versus *L* curves displayed the same generic shape consistent with the three classic stages of creep. As illustrated in figure 8, the crack propagation velocity *V* first decreases gradually (decelerating or primary creep stage) down to a minimum value (secondary creep) and then sharply increases up to failure (accelerating or tertiary creep stage). The failure time  $T_F$  is obtained by integrating dT = dL/V for *L* increasing from 0 to infinity.

358 integrating dT = dL/V for L increasing from 0 to infinity. 359  $T_F = \int_0^\infty (\sqrt{1+f(L)} - \sqrt{L})^{-n} dL$  (12)

The integral of equation 12 is convergent for all values of  $\Lambda$  satisfying the condition  $\Lambda < \Lambda_c$  and all functions 360 f(L) mentioned in previous sections. Figure 8 also shows that the decelerating stage consists of two 361 segments, first the deceleration magnitude | dV/dT | increases and then, after an inflection point is passed, 362 363 decreases down to zero (i.e., the point where the accelerating stage begins). Comparison of the right- and 364 left-hand diagrams demonstrates that increasing the polynomial degree q (for a given  $\Lambda$ ) brings the minimum propagation velocity closer to zero, increases greatly the time to failure and strongly sharpens the 365 366 transition to tertiary creep. Similar effects are produced by raising the subcritical crack growth exponent n 367 (not shown in figure 8).

Based on the results described above, three key points can be identified along the curve of V versus 368 369 T, namely, (1) the inflection point within the decelerating creep stage, (2) the minimum of the curve (note 370 that this point is the limit between the primary and tertiary creep segments; it can therefore be interpreted 371 as the center of a secondary creep segment), and (3) the failure point. These three points are distinguished 372 by their respective time coordinates,  $T_{inf}$ ,  $T_2$  and  $T_F$  (see inset in figure 8). Furthermore, in a log-log plot, the tangent at the inflection point  $T_{inf}$  defines a local power law  $V \propto T^m$ , which becomes steeper with 373 374 increasing  $\Lambda$  (see inset in figure 8). Although the individual normalized times  $T_2$  and  $T_F$  may be difficult to determine experimentally because the time normalization,  $T = t l_0/V_0$ , involves  $l_0$ , a quantity that may not 375 be experimentally accessible, the ratio  $T_{\rm F}/T_2$  is independent of normalization and can be measured in 376 377 laboratory tests.

We measured  $T_F/T_2$  for  $f(L) = (L/\Lambda)^q$  in various conditions of  $\Lambda$ , q and n. We observed that  $T_F/T_2$ strongly decreased with increasing  $\Lambda$  when q was equal to 2, changed to a much flatter, non-monotonic behavior with increasing q from 3 to 5, and eventually became a steadily but moderately increasing function of  $\Lambda$  for q = 6 (figure 9, right diagram). In all cases,  $T_F/T_2$  approached a limit value of 2 for the largest  $\Lambda$ 's while a wide range, from as high 3 to as low as 1.6, was obtained for  $\Lambda$  near zero depending on the polynomial degree q.

Because inflection points are patently difficult to determine from noisy data,  $T_{inf}$  is not a practical parameter to use for comparison with experimental data, but the exponent *m* is independent of time normalization and can be estimated as a characteristic power law exponent of rock primary creep data. We calculated *m* for a variety of values of  $\Lambda$  and *q*. We observed that *m* showed approximately logarithmic dependence on  $\Lambda$  for different values of *q* (figure 9, left diagram). The calculated *m*'s increased from values between 0.2 and 0.5, depending on *q*, at low  $\Lambda$ 's to an upper limit of about 1 at  $\Lambda$ 's approaching  $\Lambda_c$  (figure 9, left diagram).



 $\begin{array}{ccc}
392 & \Lambda & \Lambda \\
393 & Figure 9. Examples of the predicted dependence of the primary creep exponent$ *m*(left diagram) and the failure time to $394 secondary creep time ratio <math>T_{\rm F}/T_2$  (right diagram) on the characteristic flaw separation  $\Lambda$ . These curves correspond to the function 395  $f(L) = (L/\Lambda)^q$  with values of *q* indicated in matching colors (the subcritical crack growth exponent *n* was equal to 10 in all cases). 396 The positions of the limit  $\Lambda_c$  for the different exponents *q* are indicated by colored solid dots in the left diagram. See text for a 397 detailed discussion. 398

#### 3. Discussion

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#### 3.1 Comparison with experimental data: brittle creep

401 Testing this model against experimental data cannot be done directly since the essential wing crack 402 parameters *l* and *v* cannot be measured in rock samples during deformation. Even, estimating the half-length  $a_0$  of the dominant flaws from SEM images of the undeformed rock is an extremely difficult and uncertain 403 404 undertaking. We will assume here that the macroscopic creep strain rate  $\varepsilon$ ' of a very large volume of 405 material containing many dominant microcracks is linearly related to the wing crack propagation velocity 406 v. Note that the (usually non-linear) relations between v and  $\varepsilon$ ' derived in various versions of the Ashby-Hallam-Sammis model (e.g., Brantut et al., 2012) cannot be used in our model. Indeed, these relations only 407 408 include the effect of wing cracks growth and neglect the shear displacements along the dominant 409 microcracks. In our model, on the other hand, the macroscopic creep strain rate is mainly produced by 410 formation and shearing of the micro shear bands. Accordingly, the experimental equivalent of the wing 411 crack length l is the (inelastic) creep strain  $\varepsilon$ . Note that there is no measurable equivalent of  $l_0$ , making it impossible to normalize experimental time so that it can be directly compared to the model dimensionless 412 413 time  $T = t v_0/l_0$ . However, the main output parameters of the model, the exponent m and the ratio  $t_F/t_2$  can be estimated from experimental creep curves spanning the three regimes from primary to tertiary creep 414 415 without time normalization. Note that experimental data are necessarily afflicted by noise, mostly random 416 fluctuations of the readings of the measuring devices but sometimes also errors caused by computer 417 glitches. Published data sometimes contain "unphysical" features in the recorded signals, like sharp steps 418 (i.e., points of extremely high strain rates) or oscillations (i.e., alternating positive and negative strain rates), 419 which require specific removal treatments (the easiest being manually passing a smooth curve through the 420 steps or oscillations). Since the combination of regularization techniques needed for differentiating the 421 experimental  $\varepsilon$  versus t data strongly depended on the particular data being analyzed, we did not attempt to develop a comprehensive data-treatment workflow. Each data set was individually processed, although we 422 423 made every effort to maintain consistency.

424 We downloaded or digitized the published brittle creep data of Inada granite (Fujii et al., 1999), 425 Thala limestone (Brantut et al., 2013), Etna basalt (hereafter labeled Etna basalt 1; Heap, 2009; Heap et al., 426 2011) and Darley Dale sandstone (Heap, 2009; Heap et al., 2009) and calculated the corresponding time-427 dependent strain rates. In the case of Darley Dale sandstone and Etna basalt 1 (Heap, 2009), preliminary 428 constant strain rate tests at different effective confining pressures were performed to determine the rock 429 strength defined as the peak effective differential stress  $\sigma_{peak}$ - $\sigma_3$  (note that, in the following, all stresses will be understood to be effective stresses, i.e., differences of the total stresses and pore pressures). Creep tests at the same effective confining pressure (30 MPa) and various effective differential stresses  $\sigma_1$ - $\sigma_3$  below the previously measured peak stresses were then carried out in samples from the same blocks (Heap, 2009). These tests are particularly interesting to us since the samples had presumably identical properties and microstructures but were subjected to different stresses. We therefore expected that the experimental data would yield values of  $T_F/T_2$  and *m* consistent with a single value of *q* and decreasing  $\Lambda$ 's with increasing

- 436 creep stress. We also analyzed creep curves measured in basalt samples from a different outcrop on Mount
- 437 Etna (hereafter labelled Etna basalt 2, Mansbach, 2022) and a cored well in Iceland (Xing et al., 2022), and,
- in thermally cracked glass cylinders (Mallet et al., 2014, 2015).
- 439



time (min) Figure 10. Example of a strain rate versus time curve calculated from a digitized experimental creep curve (here, a Darley Dale sandstone creep test at a creep to peak stress ratio of 0.93; Heap, 2009; Heap et al., 2009). Experimental data with (dotted blue line) and without smoothing (solid black dots) are shown. The estimates of  $t_F$ ,  $t_2$  and the primary creep exponent *m* are graphically indicated. See text for a detailed discussion.

445

446 The experimental curves of  $\log(\varepsilon)$  versus  $\log(t)$  of these rock samples (e.g., figure 10) appear 447 indeed similar to the theoretical curves of figure 8. We were, therefore, able to estimate the two primary 448 output parameters,  $t_F/t_2$  and m, from these data. The time to failure  $t_F$  is easy to measure but  $t_2$  can be more 449 challenging (note that superposing the smoothed creep curves and the original noisy ones is quite helpful 450 to avoid unreasonable under- or overestimations of  $t_2$  and to estimate uncertainties). The inflection point 451 within the decelerating stage is all but impossible to identify, but the power law exponent m can still be 452 estimated by selecting a segment of data points at the center of the primary creep stage (see the example of figure 10; again superposing the smoothed and original data is a very useful precaution). Within the 453 454 estimated uncertainties, the measured values of the ratio  $t_F/t_2$  ranged from about 1.5 to 2.2, and m from 0.4 455 to slightly over 1.

456 For comparison purposes, we superposed the experimental  $(m, t_{\rm F}/t_2)$  data on the theoretical  $T_{\rm F}/T_2$ 457 versus m curves obtained by cross-plotting the numerical results for the function  $f(L) = (L/\Lambda)^q$  and constant values of q. Note that the theoretical curves converge from the left border to the vicinity of the point (m =458 459 1;  $T_{\rm F}/T_2 = 2$ ), thus delimiting a wedge-shaped region that excludes values of *m* significantly exceeding one. 460 The experimental results are in good agreement with the model in the sense that the measured data points 461 approximately fall within the allowed wedge-shaped region (figure 11). This observation also lends support 462 to our assumption that the strain rate  $\varepsilon$ ' is linearly related to the wing crack propagation velocity v. Indeed, let's assume instead that  $\varepsilon'$  is an arbitrary (monotonically increasing) function g of v. The experimental 463 464 strain rate versus time curves (e.g., figure 10) should then be compared to curves of g(V) versus T, which should have similar shapes to the curves shown in Figure 8 but yield very different values of m. These 465 466 changes would likely produce a very different Figure 11. They could significantly distort the region in (m; 467  $T_{\rm F}/T_2$ ) space allowed by the model and thus reduce or even destroy any agreement of model and 468 experiments.

Although we cannot exclude that other functions besides  $(L/\Lambda)^q$  may yield similarly satisfactory results, we can definitely eliminate the function  $A(L) = (1 + L/\Lambda)^q$  (or  $f(L) = (1 + L/\Lambda)^q$ -1), which only generated values of  $T_F/T_2$  greater than 2 that are not consistent with more than half the experimental data. We surmise that the unfitness of the function  $A(L) = (1 + L/\Lambda)^q$  is shared by all polynomial functions combining terms of widely variable degrees in  $L/\Lambda$ , including linear and quadratic terms, which produce values of  $T_F/T_2$  significantly greater than 2. If this is true, we can, hereafter, safely limit our discussion to the simple power laws  $f(L) = (L/\Lambda)^q$ .

476 If the functions  $f(L) = (L/\Lambda)^q$  are indeed the appropriate functions for interpreting rock data, we can 477 infer the values of g and A corresponding to each creep experiment. For example,  $q \approx 4$  fits both Inada 478 granite and Thala limestone although a larger  $\Lambda$  is associated to the granite than the limestone ( $\Lambda \approx 0.5$  and 479 0.1, respectively). It is tempting to interpret this result as indicating that the granite, whose low porosity 480 (0.45%) presumably consists of long, thin microcracks, has a lower flaw density than a strongly 481 heterogeneous, porous (17.5 %) carbonate. But it would be wrong to do so because  $\Lambda$  is normalized by  $l_0$ , 482 an unknown quantity that could take very different values in these two rocks. The only other observation 483 in support of taking large  $\Lambda$ 's as an indication of low flaw density is the fact that even larger values of  $\Lambda$  (> 484 1.5) correspond to the two thermally cracked glass samples. Their microstructure, indeed, exclusively 485 consists of cm-scale rather thin and smooth microcracks with intersections distant from each other by a few 486 to tens of millimeters (Mallet et al., 2014).

487 Even though the absolute values of  $\Lambda$  are practically impossible to interpret, relative variations can 488 be amenable to quantitative analysis. As mentioned earlier, creep experiments at different stress levels were 489 carried in four samples of Etna basalt 1 extracted from the same block and a similar procedure was applied 490 to three samples of Darley Dale sandstone (Heap, 2009). We therefore expect q to be constant in each rock, 491 which is indeed observed for three of the Etna basalt 1 samples (q = 2) and all the Darley Dale sandstone 492 ones ( $q \approx 7$  or 8). We also note that the inverse of  $\Lambda$  can be expressed as a second-degree polynomial in  $\sigma_1$ :

493 
$$\frac{1}{4} = C_0 - 2C_1\sigma_1 + C_2\sigma_1^2$$

Indeed, combining equations 1 to 3 yields the linear expression,  $|\tau_{\text{eff}}| = a \sigma_1 - b$ , where the positive constants are given by  $a = (1-\mu)/2$  and  $b = \sigma_3 (1+\mu)/2$ . By definition  $l_0$  is proportional to  $|\tau_{\text{eff}}|^2 = a^2 \sigma_1^2 - 2 a b \sigma_1 + b^2$ and so is  $1/\Lambda = l_0/\lambda$ , hence demonstrating equation 13. The positive constants  $C_0$ ,  $C_1$  and  $C_2$  are proportional to  $b^2/\lambda$ ,  $ab/\lambda$  and  $a^2/\lambda$ , respectively, and therefore obey the equality  $C_2/C_1 = C_1/C_0 = a/b$ .

(13)

We graphically estimated  $\Lambda \approx 0.40$ , 1.15 and 1.89 for the three Etna Basalt 1 samples with  $\sigma_1/\sigma_{peak}$ 498 499 = 0.97, 0.86 and 0.80, respectively. Using the values above, the curve of  $1/\Lambda$  versus  $\sigma_1/\sigma_{\text{peak}}$  is indeed very 500 well fitted with a second-degree polynomial of the same form as in equation 13. The estimated constants  $C_0 = 32.9, C_1 = 41.8$  and  $C_2 = 53.8$  yield ratios  $C_1/C_0 = 1.27$  and  $C_2/C_1 = 1.29$  within 1.5% of the theoretical 501 502 equality. We applied the same analysis to the Darley Dale sandstone data. A value of q greater than 6 was needed, which posed some numerical problems because we had to use a much smaller increment dL of 503 504  $3 \times 10^{-5}$  to maintain an acceptable accuracy. Using q = 8, we obtained  $\Lambda \approx 0.27$ , 0.19 and 0.059 for the creep 505 stress levels  $\sigma_1/\sigma_{\text{peak}} = 0.84$ , 0.88 and 0.93, respectively, which yielded  $C_0 = 1560$ ,  $C_1 = 1930$  and  $C_2 = 1000$ 2270, corresponding to ratios  $C_1/C_0 = 1.170$  and  $C_2/C_1 = 1.174$  in excellent agreement with the theoretical 506 507 equality. Thus, the values of  $\Lambda$  fitting the Etna basalt 1 and Darley Dale sandstone experiments are 508 quantitatively consistent with the creep stresses used in them.



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Figure 11. Experimental estimates of  $t_{\rm F}/t_2$  and m for Inada granite (black), Thala limestone (dark blue), Darley Dale 512 sandstone (light blue), Etna basalt 1 (purple), Etna basalt 2 (red), Iceland basalt (orange) and thermally cracked glass (green). The 513 error bars indicate the estimated uncertainties of the calculated values of  $t_F/t_2$  and m. In the case of Darley Dale sandstone and Etna 514 basalt 1, the creep stress levels ( $\sigma_1/\sigma_{peak}$ ) are shown above the data points in matching color. The theoretical curves of  $T_F/T_2$  versus 515 m for the function  $f(L) = (L/\Lambda)^q$  and various values of the exponent q as indicated in the inset on the right side of the diagram, are 516 superposed on the experimental data. The olive green arrow indicates the direction in which the theoretical interaction distance  $\Lambda$ 517 increases in this diagram.

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#### 3.2 Comparison with experimental data: brittle failure in constant strain rate experiments

520 Since  $f(L) = (L/\Lambda)^q$  appears to yield an appropriate description of experimental creep data in a large 521 variety of rocks, it is worth incorporating it in the failure model. Equation 6 thus becomes:

522 
$$-1.15(|\tau_{eff}| - \tau_c)\sqrt{\pi a_0 \left(1 + \left(\frac{l}{\lambda}\right)^q\right)} + \sigma_3 \sqrt{\pi l/2} = 0$$
(14)

Equation 14 can be written in dimensionless form using  $\tau^* = (|\tau_{eff}| - \tau_c)/\sigma_3$ ,  $l^* = l/(1.15^2 2a_0)$  and  $\lambda^* = \lambda/(1.15^2 a_0)$ 523 524  $2a_0$  (note that the previously used normalization of l and  $\lambda$  to  $l_0$  is not possible here because the remotely 525 applied stresses are not constant in constant strain rate experiments). Solving it for  $\tau^*$ , yields:

526 
$$\tau^* = \sqrt{\frac{l^*}{1 + \left(\frac{l^*}{\lambda^*}\right)^q}}$$
(15)

 $\tau_{\text{peak}}^* = \sqrt{\frac{\lambda^*}{q}(q-1)^{\frac{q-1}{q}}}$ 

527 Equation 15 can be used to model constant strain rate tests by calculating the variations of  $\tau^*$  associated to 528 a constant rate of increase of  $l^*$ . We thus determined the  $\tau^*$  versus  $l^*$  curves for q varying from 2 to 12 and  $\lambda^*$  from 0.1 to 16. All curves go through a maximum,  $\tau^* = \tau^*_{peak}$ , analogous to rock peak strength at  $l^* =$ 529 530  $l^*_{\text{peak}}$  (see examples in figure 12). The right-hand side of equation 15 is sufficiently simple to allow 531 determining the coordinates of the peak analytically, yielding:

$$l_{\text{peak}}^* = \lambda^* / (q-1)^{1/q}$$
(16)

533 and

532

which implies  $\tau_{\text{peak}}^* = \sqrt{l_{\text{peak}}^* (q-1)/q}$  (see derivation in appendix B). Thus,  $l_{\text{peak}}^*$  is proportional to  $\lambda^*$ 534

(17),

(the pre-factor decreasing from 1 to  $\sim 0.757$  and then slightly increasing to  $\sim 0.784$ , when q is increased 535 536 from 2 to ~4.6 and finally 8). In the same range of q,  $\tau^*_{\text{peak}}$  varies as the square root of  $\lambda^*$  (the pre-factor 537 gradually increasing from ~0.707 to ~0.828). Although the definitions of  $\lambda^*$  and  $\Lambda$  are not identical, these 538 two parameters are both related to the separation distance between microcracks and/or flaws, or, in other

words, inversely related to the flaw density. As intuitively expected, the model predicts that strength 539 540 increases with decreasing flaw density (left diagram, figure 12).



541 542

Figure 12. Examples of curves of normalized resolved shear stress  $\tau^*$  versus normalized wing crack length  $l^*$  for the function  $(l^*/\lambda^*)^q$  with q = 2 and various values of the dimensionless flaw separation  $\lambda^*$  as indicated in matching colors (left diagram). Curves of  $\tau^*$  versus  $l^*$  normalized to their peak values  $\tau^*_{peak}$  and  $l^*_{peak}$ , respectively, for various values of q as indicated in matching colors (right diagram). Importantly, these curves are independent of the values of  $\lambda^*$  used to calculate them (in other words, variations of  $\lambda^*$  at constant q yield exactly coincident curves).

548 Interestingly, normalizing the shear stress and wing crack length to their values at the peak (i.e., 549  $\tau^*/\tau^*_{\text{peak}}$  and  $l^*/l^*_{\text{peak}}$ ) produced exactly coincident curves for a given value of q, independent of  $\lambda^*$  (right 550 diagram, figure 12). Thus,  $\lambda^*$  (equivalently, the flaw density) affects the values of  $\tau^*_{peak}$  and  $l^*_{peak}$  but not the shape of the curves. Instead, it is the power law exponent q which appears to control the shape of the  $\tau^*$ 551 552 versus  $l^*$  curves, particularly the post-peak softening stage. Increasing values of q produce an increasingly 553 sharp softening post-peak behavior (right diagram, figure 12). Note that, in the softening stage,  $\tau^*$ asymptotically approaches zero while l\* increases to infinity (figure 12). This implies that formation of new 554 555 larger shear bands continues indefinitely at shear stresses ( $|\tau_{eff}|$ ) closer and closer to  $\tau_c$ , the shear stress, below which wing crack growth was not initially allowed. This property is due to the fact that the model 556 557 does not include an upper scale limit and, therefore, does not allow formation of a through-going shear band 558 like those ultimately occurring in (finite size) rock samples deformed to brittle failure.

559 Stress-strain curves measured in brittle materials are similar to the theoretical curves shown in 560 figure 12. However, unlike the  $\tau^*$  versus  $l^*$  curves, which represent the results of exclusively inelastic 561 processes, experimental stress-strain curves include both elastic and inelastic strains. For the purpose of 562 comparison with the model, the axial strain,  $\varepsilon = \varepsilon_e + \varepsilon_i$ , measured in a constant strain rate test must be 563 corrected of its elastic component  $\varepsilon_e$  so that only the inelastic strain  $\varepsilon_i$  remains. Elastic strains must obviously be dominant during the early stage of a constant strain rate test when the applied stress is too low 564 565 to produce significant inelastic deformation. This elastic stage is usually identified as the upwardly curved 566 segment, commonly observed at the beginning of the stress-strain curve (e.g., Heap and Faulkner, 2008). 567 Along this segment, the axial Young's modulus E (i.e., the slope of the stress-strain curve) increases 568 gradually owing to the closure of cracks normal to  $\sigma_1$  and reaches a maximum (i.e., inflexion point of the stress-strain curve) at  $\sigma_1 = \sigma_c$  that is generally assumed to mark the onset of inelastic deformation. The 569 570 interpretation of the upwardly curved segment as purely elastic has generally been considered satisfactory 571 in many studies where dilatancy was measured and/or acoustic emissions recorded (among others, Lockner 572 et al., 1992; Lockner, 1993; Stanchits et al., 2006; Fortin et al., 2009; Fortin et al., 2010). However, even if the purely elastic stage of a given laboratory test is accurately identified, determining  $\varepsilon_{e}$  along the rest of 573 574 the stress-strain curve cannot be done without extrapolation unless the elastic properties were actually 575 measured at regular intervals, for example, by running small cyclic stress excursions (e.g., Bernabé et al., 1994). The use of such techniques, however, is extremely rare in practice. Here, we manually digitized the 576

577 stress-strain curves measured in Etna basalt 1 and Darley Dale sandstone by Heap (2009), since we had 578 previously determined suitable values of q for these rocks, and attempted to construct models of the rocks 579 Young's modulus as a function of stress and total axial strain. Our first attempt yielded values of ( $\sigma_{\text{peak}}$  - $\sigma_c$ )/ $\sigma_3$  (i.e., the equivalent of  $\tau^*_{peak}$ ) that did not scale as the square root of  $\varepsilon_{peak}$ , the inelastic strain at the 580 peak of the stress-strain curve, as predicted by the model. However, fine-tuning the elastic model brought 581 582 the results closer to the model prediction. Since the validity of the elasticity models cannot be checked 583 independently, these efforts do not produce truly meaningful results and this approach was not pursued 584 further.

585 586

# 4. Concluding remark, implications

587 588 We developed a brittle creep and brittle failure model recognizing that flaws in rocks exist over a 589 broad range of length-scales. Our assumption behind the model is motivated by the fact that self-similarity 590 is one of the characteristics of brittle systems; grain size distributions in fault rocks (e.g. Keulen et al. 2008), 591 roughness of frictional interfaces (e.g. Candela et al. 2012), acoustic emissions recorded during experiments 592 (e.g. Goebel et al. 2017), as well as moment-magnitude scaling of crustal earthquakes (Gutenberg and 593 Richter 1944) are all suggesting that cracking is a process that is self-similar over many orders of magnitude 594 in length scale. The interactions of the dominant microcracks with smaller flaws in their vicinity, leads to 595 their coalescence, formation of micro shear bands and eventually to shear failure when the growing 596 dominant shear band reaches the sample boundaries, as typically observed in rocks deformed under 597 confining pressure. In our model, both the inclined dominant microcracks as well as their associated wing 598 cracks are allowed to grow in contrast to traditional models of brittle creep where only wing crack growth 599 is assumed (e.g. Ashby and Sammis 1990, Brantut et al. 2012). Comparison of our model to experimental 600 data suggests that the complex and non-tractable interactions of the rock microcracks and flaws can be adequately expressed by the simple power law functions,  $f(L) = (L/\Lambda)^q$ , where L is the normalized wing 601 crack length and A represents the normalized distance over which a wing crack must propagate to interact 602 603 with other flaws. The model reproduces all three characteristic stages of creep and returns experimentally 604 determinable quantities, namely the ratio of time to failure,  $t_F$ , to the time of minimum wing crack 605 propagation velocity (i.e. minimum strain rate, or center of the secondary creep segment),  $t_2$ , and a power-606 law exponent, m, that characterizes the mean deceleration rate of primary creep. The model successfully 607 fits data from a broad range of rocks and - with appropriate normalization and accounting for elastic 608 deformation - can be also used to model brittle failure. Furthermore, our model predicts that brittle creep 609 can occur over a very broad range of flaw densities and/or stress levels depending on the exact functional 610 form. Tertiary, accelerating creep has typically been observed only at a high percentage (>50%) of the ultimate failure strength (e.g. Brantut et al. 2012), however recent experiments document primary and 611 612 secondary brittle creep operating at stress levels as low as  $\sim 10\%$  of the failure strength (Xing et al. 2022) 613 providing evidence that brittle creep indeed occurs over a broad range of stress levels - resolving whether 614 creep at such low fractions of the failure strength will eventually enter the tertiary creep stage is however 615 impossible in the laboratory. As shown in figure 8, the time to failure predicted by the model varies by over 20 orders of magnitude for the variations of  $\Lambda$  explored in this work. 616

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# 5. Conclusions

We reformulated the wing crack model of brittle creep and brittle failure to allow for the formation, growthand coalescence of micro shear bands over a broad range of length scales and found that:

The model using a wide class of function A(L) properly returns classical trimodal creep curves for constant stress boundary conditions and characteristic stress-strain curves under constant strain rate boundary conditions

• The model returns experimentally determinable quantities independent of the chosen normalizations. Key outputs are the ratio of time to failure,  $t_F$ , to the time of minimum strain rate,  $t_2$ , and a power-law exponent, *m*, that characterizes the mean deceleration rate of primary creep.

- The function A(L) = 1 + f(L), where f(L) is a simple power law,  $f(L) = (L/\Lambda)^q$ , produced values of 630 the constants  $\Lambda$  and q consistent with those estimated from the experimental data. The parameter  $\Lambda$ 631 represents a normalized distance over which a flaw must propagate to interact with other flaws and 632 can be related to flaw density and stress level.
- Polynomial functions which combine the effect of multiple terms with different degrees, yielded results inconsistent with the experimental data, suggesting that brittle creep of rocks can be appropriately described using the power law function,  $f(L) = (L/\Lambda)^q$ .

#### 637 Appendix A

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The limit Λ<sub>c</sub> is the upper limit of Λ such that the curves of  $\sqrt{1 + (L/\Lambda_c)^q}$  and  $\sqrt{L}$  versus *L* are tangent. As discussed in section 2.3, the two curves do not intersect when Λ is smaller than Λ<sub>c</sub> and they intersect on two separate points for Λ strictly larger than Λ<sub>c</sub>. We now note that the curves produced by elevating the two functions above to the power 2 are also tangent to each other for  $\Lambda = \Lambda_c$ . They therefore have a single common point, where the derivatives of the functions with respect to *L* must be equal. We can therefore write the two obvious equalities  $1 + (L/\Lambda_c)^q = L$  and  $q (L^{q-1}/\Lambda_c^q) = 1$ . Eliminating *L* between these two equations yields  $\Lambda_c = (q^{1/(1-q)} - q^{q/(1-q)})^{(1-q)/q}$ .

## 646 Appendix B

647 We wish to calculate the coordinates  $\tau^*_{peak}$  and  $l^*_{peak}$  of the maximum of the curve of  $\tau^*$  versus  $l^*$ 648 predicted by equation 15. For this, we only need to calculate the derivative with respect to  $l^*$  of the right-649 hand side of equation 15:

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$$\tau'(l^*) = \frac{1 + \left(\frac{l^*}{\lambda^*}\right)^q - q\left(\frac{l^*}{\lambda^*}\right)^q}{2\sqrt{l^*} \left(1 + \left(\frac{l^*}{\lambda^*}\right)^q\right)^{3/2}}$$

651 Solving  $\tau'(l^*) = 0$  for  $l^*$  yields the solution  $l_{\text{peak}}^* = \lambda^*/(q-1)^{1/q}$  and plugging this value in equation 15

652 produces 
$$\tau_{\text{peak}}^* = \sqrt{\frac{\lambda^*}{q}(q-1)^{\frac{q-1}{q}}}$$
, which can be rewritten  $\tau_{\text{peak}}^* = \sqrt{l_{\text{peak}}^*(q-1)/q}$ .

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#### 654 Data Availability Statement

Our model is analytical. Computer programming is therefore unnecessary in principle. However certain parameters such as the power law exponent *m* are easier to determine numerically. To help interested readers, the Mathematica script used to construct and interpret Figure 8 is available at Zenodo. Except for the Iceland and Etna 2 basalts, we used published experimental data that can be obtained from the articles cited. The Iceland and Etna 2 basalts data were produced in our laboratory and are available at Zenodo at 10.5281/zenodo.6463941.

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